KINETIC AND EQUILIBRIUM DATA OF GASES IN GLASS MELTS

MARCELA JEBAVÁ, LUBOMÍR NĚMEC, JAROSLAV KLOUŽEK

Laboratory of Inorganic Materials Joint Workplace of the Institute of Inorganic Chemistry of the Academy of Sciences of the Czech Republic and the Institute of Chemical Technology Prague Technická 5, 166 28 Prague, Czech Republic

E-mail: nemec@iic.cas.cz

Submitted June 23, 2004; accepted September 1, 2004

Keywords: Mathematical model, Glass, Bubbles, Gas diffusion

The method of the high temperature observation with subsequent bubble analysis by mass spectrometry was used for the determination of physical constants of gases dissolved in glass melts. The procedure arises from the mathematical model of the bubble behaviour in glass melt, namely the bubble size and composition developments. These two time dependences were also experimentally measured and used for the acquisition of diffusion coefficients, physical solubilities and actual concentrations of CO_2 , N_2 and Ar in the TV glass at temperatures 1300°C and 1400°C.

INTRODUCTION

Solubility and diffusion of gases in glass melts are associated with several significant glass areas such as glass structure, glass permeability for gases or glass technology. The interactions between glass melts and gases influence particularly the melting process of glass and resulting glass colour. The presence of gases in glass melts has both chemical and physical nature, but in most cases the physical solubility of gases and their diffusion coefficients in melt are data needed for quantitative description of bubble behaviour in glass melts and oxidation-reduction interactions of gases in glass melts determining both refining and glass colour, as well as glass heating and its workability. The application of rather complicated mathematical models of the melting process thus considerably depends on the accuracy of physical solubility and diffusion coefficients of appropriate gases in glass melts.

The methods of diffusion coefficient measurement in glass melts are mostly based on following the interaction between defined volume of gas and glass melt. Bubbles containing the examined gas or diffusion vessels with gas, immersed in the melt, were applied to measure the gas volume dissolution in the melt [1,2]. To acquire the appropriate diffusion coefficient, the physical solubility of the gas in the melt was nevertheless needed. No other gas was allowed in the melt to prevent its impact on mass transfer. The physical solubility of gases in glass melts is in relation to the free volume in the glass structure, available for molecules of dissolved gas [3]. The values of the Ostwald's absorption coefficient of H_2 , He, Ne and O_2 range from 0.01 to 0.03 [4,5] and are only slightly dependent on temperature. Then the approximate value of gas solubility in glass melt varies in the range of tenths of moles per m³ of glass melt. The very low value of physical solubility is however determined with high inaccuracy by current analytical methods. That is why we seek for new approaches to the determination of both diffusion coefficients and solubilities of gases in glass melts [6].

The goal of this work is to develop the procedure leading to the acquisition of diffusion coefficients and physical solubilities of technologically important gases in glass melts. The mathematical model of multicomponent bubble and the method of visual bubble observation in the melt are applied to acquire the above-mentioned values. The primary aim of the work is to facilitate the mathematical modeling of bubbles during melting process under real conditions.

THEORETICAL

The kinetic equations describing the single bubble behaviour involve the term $D_i^{2/3}(m_{ib} - m_{ia}) = D_i^{2/3}(m_{ib} - L_ip_i)$ where D_i is the diffusion coefficient of the i-th gas, m_{ib} and m_{ia} are gas concentrations in bulk glass and on the bubble surface. The quantity L_i is the physical solubility of the i-th gas in the melt and p_i is its partial pressure in the bubble. If the sulphate layer is present on the bubble surface, the driving force for the gas transfer can be expressed by $k(p_{so2}p_{o2}^{1/2} - p_{so2s}p_{o2s}^{1/2})$ where k is the rate constant of the layer formation or decomposition and p_i and p_{is} are the partial pressure and saturation partial pressure of SO₂ and O₂ in the bubble.

Neglecting the formation of the sulphate layer, the set of equations describing the bubble behaviour at a constant temperature for n diffusing gases has the form [7]:

$$\frac{da}{d\tau} = \frac{2g^2 \rho^2 a^3}{27\eta p_{ex}} + \frac{A}{p_{ex}} \sum_{i=1}^n \frac{D_i^{2/3}}{m_i} (m_{ib} - L_i p_i)$$
(1)

$$\frac{dp_i}{d\tau} = \frac{3A}{a} \frac{D^{2/3}}{M_i} + (m_{ib} - L_i p_i) \frac{3p_i}{a} \frac{da}{d\tau} \qquad (i = 1, 2... n - 1)$$
(2)

$$p_i = \sum_{i=1}^n p_i \tag{3}$$

where $A = 0.382RTg^{1/3}\rho^{1/3}/\eta^{1/3}$, *a* is the bubble radius, p_t is the total pressure in the bubble, M_i is the molecular mass of the i-th gas and ρ and η are glass density and viscosity.

The constant to be determined in equations (1) and (2) are D_i , m_{ib} and L_i . As the mentioned quantities always occur in the form of products $D_i^{2/3}m_{ib}$ and $D_i^{2/3}L_i$, equations (1) and (2) formally involve only two unknown parameters, namely $D_i^{2/3}m_{ib}$ and $D_i^{2/3}L_i$.

The experimental measurements usually provide bubble size development $a = a(\tau)$ and bubble composition development $p_i = p_i(\tau)$. After fitting both experimental dependences by empirical functions, the derivatives $da/d\tau$ and $dp_i/d\tau$ can be obtained. If the experimental values of $da/d\tau$ and $dp_i/d\tau$ are measured for the same or at least very similar bubble, the unknown products, $D_i^{2/3}m_{ib}$ and $D_i^{2/3}L_i$, can be calculated from equations (1) and (2). In order to obtain the mentioned values, at least two experimental points must be known for both $a = a(\tau)$ and $p_i = p_i(\tau)$ dependences.

As equation (1) includes the unknown parameters of all diffusing gases, the following arrangement should be performed.

The term $1/M_i(D_i^{2/3}m_{ib} - D_i^{2/3}L_ip_i)$ is expressed from equation (2) for all gases with the exception of the gas whose properties are to be determined:

$$(D_i^{2/3} m_{ib} - D_i^{2/3} L_i p_i) = \frac{a}{3A} \left(\frac{dp_i}{d\tau} + \frac{3p_i}{a} \frac{da}{d\tau} \right)$$
(4)

The values of $1/M_i(D_i^{2/3}m_{ib} - D_i^{2/3}L_ip_i$ for (i = 1, 2..., n - I) are substituted into equation (1):

$$\frac{da}{d\tau} = \frac{A}{p_i} \left\{ \sum_{i=1}^{n-1} \frac{a}{3A} \left[\frac{dp_i}{d\tau} + \frac{3p_i}{a} \frac{da}{d\tau} \right] + \frac{1}{M_i} \left(D_i^{2/3} m_{ib} - D_i^{2/3} L_i p_i \right) \right\}$$
(5)

Then the appropriate form of equation (2) is:

$$\frac{dp_i}{d\tau} = \frac{3A}{aM_i} \left(D_i^{2/3} m_{ib} - D_i^{2/3} L_i p_i \right) - \frac{3p_i}{a} \frac{da}{d\tau}$$
(6)

where *i* is the index of the examined gas.

Taking into account two arbitrary chosen experimental points, namely j and j+1, the values of the products D_i , m_{ib} and L_i (the properties of the examined gas) can be explicitly expressed by two equations. Starting from equation (6), we will get:

$$D_i^{2/3} L_i = \frac{a_{j+1}M_i}{3A(p_{ij} - p_{ij+1})} \left[\frac{dp_{ij+1}}{d\tau} + \frac{3p_{ij+1}}{a_{i+1}} \frac{da_{j+1}}{d\tau} - \frac{a_j}{a_{j+1}} \frac{dp_{ij}}{d\tau} - \frac{3p_{ij}}{a_{j+1}} \frac{da_j}{d\tau} \right]$$
(7)

$$D_i^{2/3}m_{ib} = \frac{a_j M_i}{3A} \frac{dp_{ij}}{d\tau} + D_i^{2/3} L_i p_{ij} + \frac{M_i p_{ij}}{A} \frac{da_j}{d\tau}$$
(8)

As it is obvious from equation (7), the relation is applicable only when $p_{i,j} \neq p_{i,j+1}$, i.e. in the stage of developing bubble composition. To get separated values of D_i , m_{ib} and L_i from both products, one value should be obtained by another independent method. During later stages of bubble life, its composition is almost constant and the bubble growth is almost linear (bubble stationary state). Starting from equation (5) or (6) for $dp_i/d\tau$, $dp_i/d\tau = 0$ we have after integration:

$$a = a_0 + \frac{A}{p_i M_i} \left(D_i^{2/3} m_{ib} - D_i^{2/3} L_i p_i \right) \tau$$
(9)

where *i* is the index of the examined gas. If follows for two experimental points, *j* and j+1:

$$D_i^{2/3} m_{ib} - D_i^{2/3} L_i p_{ij} = \frac{p_i M_i}{A} \frac{a_{j+1} - a_j}{\tau_{j+1} - \tau_j}$$
(10)

As it is clear from equation (10), only the value of the term $D_i^{2/3}m_{ib} - D_i^{2/3}L_ip_i$ can be obtained from the arbitrary two experimental points *j* and *j*+1 lying on curves $a = a(\tau)$ and $p_i = p_i(\tau)$. In order to obtain all three parameters, namely D_i , m_{ib} and L_i , two of them should be acquired by other methods.

Verification of the empirical procedure

The experimental dependences $a = a(\tau)$ and $p_i = p_i(\tau)$ should be replaced by empirical ones in order to get the values of appropriate derivatives, i.e. $da/d\tau$ and $d_{pi}/d\tau$. This step may be risky as the values of $da/d\tau$ and $d_{pi}/d\tau$ are very sensitive to the course of empirical functions. That is why the verification of the procedure was undertaken by using experimental tuned values of gas properties being summarized in table 1. These data were obtained from the behaviour of air and CO₂ bubbles.

The proposed empirical functions respecting the physical nature of bubble behaviour have the form:

$$a = (a_0 - a' - k_1 \tau) \exp(-k_2 \tau^{k_1}) + a' + k_1 \tau$$
(11)

$$p_i = (p_0 - p_{ex}) \exp(-k_1 \tau^{k_i}) + p_{ex}$$
 (12)

where a', k_1 , k_2 and k_3 are empirical constants.

The verification procedure consisted of following steps:

- 1. The experimental tuned values of D_i , m_{ib} and L_i were fed into the mathematical model of bubble behaviour (eqs. 1-3) and values of $da/d\tau$ and $dp_i/d\tau$ were calculated.
- 2. The relations $da/d\tau$ and $dp_i/d\tau$ were replaced by appropriate empirical functions (11) and (12) and the new values of *a* and *p_i*, as well as $da/d\tau$ and $dp_i/d\tau$ were acquired as a function of time.
- 3. The values of products $D_i^{2/3}m_{ib}$ and $D_i^{2/3}L_i$ were calculated from equations (7) and (8) and subsequently, the values of D_i and L_i were acquired by using the original value of m_{ib} .
- 4. The new set of values of D_i , m_{ib} and L_i was used in the mathematical model to calculate new courses of $a = a(\tau)$ and $p_i = p_i(\tau)$.

The courses of $a = a(\tau)$ and $p_i = p_i(\tau)$ from points 1 and 4 are compared in figures 1 and 2. While the composition developments exhibit excellent agreement, the deviation between both courses of $a = a(\tau)$ dependences arises in later stages. The deviation coming from difference between original and empirical equations is acceptable up to about 600 min, i.e. longer than are bubble lifetimes under real conditions.

Table 1. The comparison of experimental tuned values of CO_2 and N_2 constants and those obtained from empirical procedure. Initial bubble compositions 100% CO_2 and 100% N_2 , at 1200°C.

		experimental tuned	values resulting from empirical procedure	
		values	100% N ₂	100% CO ₂
CO_2	D_i (m ² /s)	3.818×10^{-14}	3.430×10^{-14}	4.090×10^{-14}
	L_i (kg/m ³ Pa)	$4.588 imes 10^{-6}$	4.420×10^{-6}	4.660×10^{-6}
	m_{ib} (kg/m ³)	$9.800 imes 10^{-2}$	9.800×10^{-2}	9.800×10^{-2}
N ₂	D_i (m ² /s)	1.202×10^{-11}	1.250×10^{-9}	2.200×10^{-11}
	L_i (kg/m ³ Pa)	1.919×10^{-9}	1.530×10^{-9}	2.490×10^{-8}
	m_{ib} (kg/m ³)	$1.500 imes 10^{-4}$	1.500×10^{-4}	1.500×10^{-4}

Table 2. The comparison of artificial values of CO_2 and N_2 constants and those obtained from empirical procedure. Initial bubble compositions 100% CO_2 and 100% N_2 , at 1200°C.

		artifical	values resulting from empirical procedure	
		values		
			100% N ₂	100% CO ₂
	D_i (m ² /s)	$5.00 imes 10^{-13}$	5.08×10^{-13}	5.11×10^{-13}
CO_2	L_i (kg/m ³ Pa)	2.00×10^{-6}	$1.97 imes 10^{-6}$	1.99×10^{-6}
	m_{ib} (kg/m ³)	1.00×10^{-1}	1.00×10^{-1}	1.00×10^{-1}
N ₂	D_i (m ² /s)	1.00×10^{-12}	0.97×10^{-12}	1.08×10^{-12}
	L_i (kg/m ³ Pa)	$1.00 imes 10^{-6}$	0.99×10^{-6}	$0.99 imes 10^{-6}$
	m_{ib} (kg/m ³)	5.00×10^{-2}	5.00×10^{-2}	5.00×10^{-2}

Ceramics - Silikáty 48 (3) 121-127 (2004)

Experimentally tuned N_2 and CO_2 data however exhibit non-probable high mutual differences. Therefore we apply also the set of original artificial values estimated from the molecular size of respective gases. Table 2 illustrates that the application of the empirical procedure brings excellent values of gas constants differing from the original artificial ones within 2 %.



Figure 1. The comparison of courses of bubble size development using original experimental tuned data with those obtained by the empirical procedure.



Figure 2. The comparison of courses of bubble composition developments using original experimental tuned data values with those obtained by the empirical procedure.

EXPERIMENTAL

The method of high temperature observation of gas bubble in a glass melt was used to get the experimental dependences $a = a(\tau)$ and $p_i = p_i(\tau)$. The TV glass used for the study was prepared from dried materials of p.a. purity to get a negligible amount of chemically dissolved gases. The glass sample was inserted into a flat silica glass observation cell and heated to the measuring temperature in the laboratory furnace with an observation window. After glass melting and refining, a small bubble of defined initial composition was created by the silica glass tube with inlet of defined gas and immersed in glass. The bubble was subsequently "shuttled" in the glass melt by sucking and expelling the bubble into or out of the silica glass tube as is schematically presented in figure 3.



Figure 3. Scheme for the "shuttle" method.

The bubble size development was continually recorded by a digital camera. Figure 4 presents an example of a high temperature image. After finishing the experiment, the small glass sample with the examined bubble was taken out of the melt and cooled. The bubble composition was determined by mass spectrometry.

The experiments were performed with bubbles initially containing 100% Ar at temperatures 1300°C and 1400°C. The high temperature observations of bubble size development followed by the bubble composition analysis were made at each temperature for 4-7 time expositions.

RESULTS

Figure 5 shows a typical experimental bubble size development and the course of empirical function (equation 11). As it is obvious from this figure, at least one from the couple of points necessary to calculate products $D_i^{2/3}m_{ib}$ and $D_i^{2/3}L_i$ (see equations 7 and 8)



Figure 4. Micrograph of high temperature observation of a bubble in the laboratory furnace.



Figure 5. Experimental bubble size development and the course of the empirical function at temperature 1400°C.

should come from the time interval between 0 and about 2000 s. Another set of experimental results involved bubble analyses after finishing bubble observations at high temperatures. The experimental dependences of partial pressures of single gases on time, supplemented by empirical functions (equation 12) are presented in figure 6a-c.

The compositions of bubbles at a given time should be recalculated to the reference diameter using the equation:

$$c = \frac{d_0}{d} c_0 \tag{13}$$

where *c* and *d* are concentration of given gas and diameter of measured bubble, and c_0 and d_0 are the same quantities relating to the reference bubble.

The values of both diameters and partial pressures as well as appropriate derivatives, namely $dd/d\tau$ and $dp_i/d\tau$, were acquired from empirical functions and the products of $D_i^{2/3}m_{ib}$ and $D_i^{2/3}L_i$ were calculated using equations (7) and (8). Table 3 presents average values of the above-mentioned products together with the standard relative deviation. The values of products were used to calculate the developments of bubble diameter and composition in time, i.e. to simulate the behaviour of experimental bubbles using the obtained data.

Figure 7 brings the comparison of experimental points and the calculated values of the bubble size at temperature 1300°C. The appropriate experimental and calculated values of bubble compositions are presented in figures 8 (1300°C) and 9 (1400°C).

Table 3. The average values of products and their standard deviation at the temperatures 1300° C and 1400° C.

	1300°C		
	$D_i^{2/3}L_i$	$D_i^{2/3}m_{ib}$	
CO_2	$1.41 \times 10^{-14} \pm 1.89 \times 10^{-15}$	$1.11 \times 10^{-9} \pm 6.34 \times 10^{-11}$	
N_2	$1.64 \times 10^{-14} \pm 2.45 \times 10^{-15}$	$8.48 \times 10^{-11} \pm 8.87 \times 10^{-12}$	
Ar	$1.22 \times 10^{-14} \pm 1.17 \times 10^{-15}$	$6.25 \times 10^{-10} \pm 7.59 \times 10^{-11}$	
1400°C			
	$D_i^{2/3}L_i$	$D_i^{2/3}m_{ib}$	
CO_2	$1.04 \times 10^{-14} \pm 7.13 \times 10^{-16}$	$1.40 \times 10^{-9} \pm 3.08 \times 10^{-11}$	
N_2	$2.89 \times 10^{-15} \pm 1.35 \times 10^{-15}$	$8.72 \times 10^{-11} \pm 6.47 \times 10^{-12}$	
Ar	$2.56 \times 10^{-14} \pm 2.02 \times 10^{-15}$	$1.03 \times 10^{-9} \pm 9.22 \times 10^{-11}$	

DISCUSSION

The values of products $D_i^{2/3}m_{ib}$ presented in table 3 increase with temperature for all examined gases. This tendency confirms the theoretical expectation. As the bulk gas concentration m_{ib} is supposed independent on temperature, the increase of the product with temperature is affected by an increase of the appropriate diffusion coefficient.



Figure 6. The experimental bubble composition developments simulated by empirical functions at temperature 1400°C. a) CO_2 , b) N_2 and c) Ar.

The product $D_i^{2/3}L_i$ may exhibit a complicated temperature dependence. In case of CO₂, the decrease of the product with temperature corresponds to the relatively steep decrease of CO₂ solubility [8]. Only physically dissolved gases, namely N₂ and Ar, should exhibit slight solubility increase with temperature as a consequence of thermal expansion of glass structure. However, only the argon product shows the expected temperature increase. In addition, the product $D_{N2}^{2/3}L_{N2}$ brings the highest relative experimental deviation 47 % while the same deviation for Ar and CO₂ is up to 10 %.



Figure 7. Experimental and calculated courses of bubble size developments at temperatures 1300°C and 1400°C.



Figure 8. Experimental and calculated courses of bubble composition developments at temperature 1300°C.



Figure 9. Experimental and calculated courses of bubble composition developments at temperature 1400°C.

Nevertheless, the use of products shows a good agreement if applied for simulation of experimental courses $p_i(\tau)$ as is obvious from figures 8 and 9. The comparison of bubble size developments, obvious from figure 7, shows a growing deviation of the model from the experiment; however, the relative error after 15600 s did not exceed 10 %. Thus, the values of products appear suitably precise for the mathematical simulation of bubbles. This fact facilitates the mathematical modeling of bubbles sources under real conditions with an acceptable precision.

As it was mentioned in the Theoretical, separated values of gas properties D_i , m_{ib} and L_i can be obtained from corresponding products when one gas quantity is acquired by another independent method. In this study, the measured values of CO₂ bulk concentration and estimated quantities of Ar and N₂ were applied [8].

Unfortunately, only values for CO₂ may be considered to correspond to the real ones. The gas solubility of tenths of mols per m³ of glass is a rasonable value, as well as the values of the diffusion coefficient when comparing them with corresponding values in float glass [8]. However, the only estimated actual concentrations of both Ar and N₂ in the melt devaluated the appropriate values of diffusion coefficients and solubilities. As is obvious from table 4, especially the values of D_{Ar} are too high and L_{Ar} too low with respect to the expected values. The more reliable procedure is therefore needed to determine one of three quantities by an independent method. As gas solubilities in glass melts differ much less than their bulk concentrations in glass, the physical solubility measurement may become the required independent measurement.

Table 4. The calculated values of CO₂, N₂, Ar properties and experimentally obtained bubble growth rates $da/d\tau$ at the temperatures 1300°C and 1400°C.

		1300°C	1400°C
	D_i (m ² /s)	1.20×10^{-12}	1.70×10^{-12}
CO_2	L_i (kg/m ³ Pa)	1.24×10^{-6}	7.79×10^{-7}
	m_{ib} (kg/m ³)	9.80×10^{-2}	9.80×10^{-2}
	D_i (m ² /s)	4.27×10^{-10}	$4.44 imes 10^{-10}$
N ₂	L_i (kg/m ³ Pa)	2.89×10^{-8}	4.82×10^{-9}
	m_{ib} (kg/m ³)	1.50×10^{-4}	1.50×10^{-4}
	D_i (m ² /s)	$4.44 imes 10^{-8}$	9.40 × 10 ⁻⁸
Ar	L_i (kg/m ³ Pa)	9.81×10^{-10}	1.24×10^{-9}
	m_{ib} (kg/m ³)	$5.00 imes 10^{-5}$	5.00×10^{-5}
	$da/d\tau$ (m/s)	5.04×10^{-9}	1.11 × 10 ⁻⁸

CONCLUSIONS

ment of bubble size and composition developments was

used for the acquisition of diffusion coefficients, solu-

bilities and bulk concentrations of CO2, N2 and Ar in TV

of artificial bubbles is usable for the mathematical si-

mulation of bubbles in melts, but another independent

and sufficiently precise method is missing to get suffi-

ciently true values of physical gas constants in glass

melts, namely their diffusion coefficients, solubilities

mentioned procedure for other technologically impor-

tant gases, i.e. O₂, SO₂ and H₂O. This is an object of

The question arises about the applicability of the

glass at temperatures 1300°C and 1400°C.

and actual concentrations in the melt.

authors' next effort.

The procedure based on the experimental measure-

The presented results show that the applied method

References

- 1. Doremus R. H.: J.Am.Ceram.Soc. 49, 461 (1966).
- 2. Němec L., Kloužek J.: Ceramics-Silikáty 39, 1 (1995).
- Doremus R. H.: *Glass Science*, p.122, John Wiley & Sons Inc., New York 1994.
- Perkins W. G., Begael D.R.: J.Chem.Phys. 54, 1683 (1971).
- 5. Norton F. J.: Nature 191, 701 (1961).
- 6. Ramos J. I.: J.Am.Ceram.Soc. 69, 149 (1986).
- Němec L., Kloužek J.: Ceramics-Silikáty 47, 81 (2003).
- 8. Kloužek J.: Unpublished results

KINETICKÁ A ROVNOVÁŽNÁ DATA PLYNŮ V ROZTAVENÝCH SKLECH

MARCELA JEBAVÁ, LUBOMÍR NĚMEC, JAROSLAV KLOUŽEK

Laboratoř anorganických materiálů, společné pracoviště Ústavu anorganické chemie AV ČR a Vysoké školy chemicko-technologické v Praze Technická 5, 166 28 Praha

Acknowledgement

This work was supplied with the subvention by the Grant Agency of ASCR, Project No. S4032103 and by the Ministry of Education, Youth and Sports of the Czech Republic, Project No. MSM 223100002.

Metoda vysokoteplotního sledování s následnou analýzou bublin hmotnostní spektrometrií byla použita pro stanovení vlastností plynů v roztavených sklech. Metoda vychází z matematického modelu chování bublin v tavenině simulujícího časový vývoj jejich rozměru a složení. Obě závislosti byly experimentálně měřeny a využity ke stanovení difúzních koeficientů, rozpustností a aktuálních koncentrací CO₂, N₂ a Ar ve skle pro televizní obrazovky při teplotách 1300°C a 1400°C.