THE CHARACTERIZATION OF THE Ca–K GEOPOLYMER/SOLIDIFIED FLUID FLY-ASH INTERLAYER

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A Ca–K geopolymer matrix based on clay material and blast-furnace slag was filled with aggregates, ash pellets made from fluid fly ash, and the interlayer formed between the two components was studied. The scanning electron-microscopy investigation of the inseparable interlayer demonstrated that the pellets were not only enveloped in a geopolymer matrix but also incorporated through a thin, yet identifiable, surface pellet layer. The migration of calcium and potassium ions was detected and that changes in the quantity of these ions arise from their mobility. The interlayer on the edges of ash pellets was also studied by infrared analysis, which in this layer proved bands belonging to both participants, the matrix and the pellets. Based on the results, two different materials prepared from wastes could be used for the preparation of a new composite material and thus facilitate waste-material disposal.

INTRODUCTION

The amount of ash from coal-burning power plants rises significantly every day despite the fact that new effective technologies of coal combustion have been invented. Currently, produced ashes are typically dumped on landfills, and only a small part of these ashes is used [1-3], often on a laboratory scale only, e.g. for zeolite synthesis [4], construction materials [5-6], alumina extraction [7], glass–ceramic production [8], geopolymer matrix preparation [9-10] or as a road-base material [11].

The fluidized bed combustion (FBC) of coal minimizes air pollution caused by sulfur oxides which makes it environmentally friendly. A specific FBC temperature yields a combination of two main groups of ash components. One of them results from the addition of calcium carbonate into the burning layer, while the other is a residue of predominantly clayed materials. A fluid-burning temperature at 820°C decomposes the calcium carbonate, and the resulting calcium oxide captures almost all sulfur oxides from the coal and forms an anhydrite (CaSO₄). On the other hand, the crushed-coal particles burn in the fluid layer, and their effective burning is ensured by the revolve-ring system, where partially burned particles return to the fluid layer [11-12]. The clayed residue in coal is affected by the elevated temperature, and the revolve-ring burning system offers time for the rearrangement of the aluminum coordination to the oxygen in an alumina-silicate clayed lattice, which thereby produces changes in the lattice [13-15].

Previous works on the solidification of fluid coal ashes (C-ashes according to the ASTM classification) have demonstrated the formation of a solid, insoluble material when ash is treated (e.g. milled) and adequately watered [16-17]. Five-year experiments have verified their long-term stability even when they were submerged in water. The obtained slightly grayish cementing matter could be filled with various types of fillers (quartz sand, blast-furnace slag, etc.).

Understanding fluid-ash solidification was only the first step in the process of reusing these ashes. The second step was to find a way to use the solidified matter and take advantage of its durability, mechanical properties, and insolubility in water. The solidified fluid fly ash (FFA) was shaped into rounded oval pellets (12 mm in diameter), which were used as aggregates when surrounding the Ca–K geopolymer matrix. The stability of the obtained material is closely connected to the formation of a stable coating interlayer between the two participants.

This article presents the characterization of the interlayer formed in a new composite material from a Ca–K geopolymer matrix and pellets based on solidified fluid fly ash by scanning electron microscopy and infrared analysis.
MATERIALS AND METHODS

Materials

Ash pellets

Previously obtained results based on experiments with the solidification of fluid coal ashes proved the formation of a solid and insoluble material with compressive strength between 13.0 and 15.0 MPa [16-17]. The knowledge of the stability of the material when submerged in water and the mechanical properties [16] were the main reasons for using this material as an aggregate in the form of pellets.

The ash pellets were prepared from fluid fly ash (Kladno Power Plant, Central Bohemian Region) in a mixture with bentonite. The best result was obtained with the combination of 10 wt. % of bentonite mixed into the fluid fly ash [16]. This clayed material comes from the northern part of Bohemia and is supplied by the company Keramost a.s., Most (Czech Republic). The main role of the natural milled bentonite is to fill the space between the round-shaped fly-ash particles. Moreover, it is well-known in ceramic practice that it also serves as a plasticizer when mixed with ash and water.

The chemical compositions of relevant materials are shown in Table 1, including loss-on-ignition (LOI) values. Table 2 presents the X-ray powder diffraction (XRD) results of the two raw materials needed for ash-pellet preparation.

The results have shown that fresh fluid fly ash (FFA) has a major content of anhydrite, lime and quartz and a minority of calcite and hematite. Due to a reaction of FFA with air humidity, portlandite (Ca(OH)$_2$) could sometimes be detected instead of lime. Bentonite is formed by montmorillonite, quartz, anatase and calcite, with a minor amount of muscovite.

A commonly used pellet machine was applied for the preparation of ash pellets. The most important factors are the mixing time and an adequate proportion of water. The technology used requires 6.5 - 8 wt. % of water. The pellets were stored under the cover of a plastic sheet to prevent fast water evaporation.

A geopolymer matrix

The geopolymer matrix (the Ca–K geopolymer binder) was based on a mixture of an alkalized clayed material and blast-furnace slag in the ratio of 100:70,
studied and described earlier [18-20]. The geopolymer matrix components were: \(K_2O/Al_2O_3 = 0.75; SiO_2/Al_2O_3 = 2.96; H_2O/K_2O = 12.42\). The clayed material was Mefisto, an industrially prepared material with a metakaolin base (ČLUZ, Nové Strašecí, Czech Republic). The blast-furnace slag originated from a Kladno deposit (Central Bohemia); it was a by-product of cast-iron production in the 19th and 20th centuries. Tables 3 and 4 show their chemical and mineralogical compositions, respectively. The X-ray powder diffraction (XRD) of Mefisto has identified, in addition to several crystalline phases listed in Table 4, a large amount of an amorphous phase originating from the decomposition of kaolinite lattice during the industrial thermal activation of the clayed material. The dominant components in the blast-furnace slag (Table 4) are gehlenite (\(Ca_2Al(Si_2O_7)\)) and calcite (\(CaCO_3\)) with minor participation of wolastonite merwinite, bassanite and syngenite.

Sample preparation

The dried ash pellets, used as pebbles, were placed into a plastic box and cemented together with a freshly prepared Ca–K geopolymer binder (the clay material Mefisto was stirred with the alkaline solution for 10 min., after which the blast-furnace slag was added and mixed for another 10 min). The geopolymer matrix was poured into these scattered pellets (a cross-section can be seen in Figure 1). The pellets, as aggregates, were submerged into the Ca–K geopolymer matrix. The high viscosity of the matrix caused the formation of some enclosed air bubbles. On the left side, we could see slightly enveloped pellets, although they were firmly fixed to the sample surface.

![Figure 1. A cross-section of the Ca–K geopolymer filled with FFA pellets.](image)

Methods

Non-destructive X-ray fluorescence (XRF) spectroscopy (Spectro IQ, Kleve, Germany) was used. The target material was palladium and the target angle from the central ray was 90°. The focal point was a 1 mm × 1 mm square, with a maximum anode dissipation of 50 watts and 10 cfm forced-air cooling. The instrument was equipped with a HOPG Barkla crystal. The tested samples were prepared using the pressed-pellet method: 4.0 g of the material (a particle size of 15 - 20 µm) was mixed for 10 minutes with 0.9 g of a binding additive (HWC Hoechst wax, Germany). The pressing power was 80 kN. All of the results were automatically recalculated to the oxide forms.

The X-ray powder diffraction (XRD) data were collected at room temperature with an X’Pert PRO 0-0 powder diffractometer with parafocusing Bragg-Brentano geometry using CuKα radiation (\(λ = 1.5418 \text{ Å}, U = 40 \text{ kV}, I = 30 \text{ mA}\)). The data were scanned with an ultrafast detector X’Celerator over the angular range of 5 - 60° (2θ) with a step size of 0.0167° (2θ) and a counting time of 20.32 s step⁻¹. Data evaluation and identification of mineral phases were performed in the software package HighScore Plus.

The sample surface was studied using scanning electron microscopy (SEM) and energy dispersive spectroscopy (EDS) for an X-ray microanalysis. The sample was Au-coated with a K550X Emitech sputter coater in an atmosphere of argon and was examined with a Quanta 450 scanning electron microscope (EDAX Company, USA). The high vacuum mode was used, and images were taken with an ETD (Everhart-Thornley detector) in the secondary electron mode. Using the same microscope, an X-ray microanalysis (EDS analysis) of the chemical elements was conducted with an EDS silicon drift Si (Li) Apollo detector and an EIT preamplifier. Data acquisition was performed using EDAX TSL OIM software with ZAF corrections.

The infrared (IR) spectra were measured using a Protégé 460 E.S.P. (Thermo Nicolet Instruments Co., Madison, USA) spectrometer over the range of 4000 - 400 cm⁻¹ at a resolution of 4 cm⁻¹, averaging 128 scans by the ATR technique to minimize disruption to the sample caused by the preparation techniques [21]. Spectral decomposition (or deconvolution) was performed using the OMNIC 7 program with a Gaussian function. The starting calculation parameters for the curve-fitting process, such as the number of component bands, band positions, width at half-maximum, relative intensities, and shape, were predetermined by the combined procedures of the secondary derivative method and Fourier self-deconvolution.

RESULTS AND DISCUSSION

XRD results

A geopolymer matrix

The combination of the clayed material and blast-furnace slag results in a geopolymer that can be referred to as a Ca–K geopolymer matrix. XRD analysis of the resulting mixture confirms a roentgen-amorphous phase and the direct participation of the blast-furnace slag in the constitution of the final geopolymer matrix, as
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presented in Table 5. The content of gehlenite is reduced considerably, and new phases appear: arcanite (K$_2$SO$_4$), calcite (CaCO$_3$), and andradite (Ca$_3$Fe$_2$(SiO$_4$)$_3$).

**Ash pellets**

The mineralogical composition of ash pellets, a material originating from fly ash (FFA) and bentonite, is a combination of roentgen-amorphous alumino-silicates and calcium carbonates (calcite/aragonite (CaCO$_3$)), bassanite (CaSO$_4$·0.5H$_2$O) and quartz (SiO$_2$) complemented with hematite (Fe$_2$O$_3$) and other phases in trace amounts.

**SEM characterization**

SEM and EDS studies of the polished specimen from the sample show a heterogeneous calcareous substance in the previously described pellets [16-17], in which the ash alumino-silicates form short- or longer-chained structures surrounded by a dominantly calcareous matter. Figure 2 depicts the coating material directly bonded to the pellets; its color changes are reflected in the various shades of the geopolymer binder at certain distances from the pellet surfaces.

The greater the distance from the pellet edges, the larger the change in the shade from whitish color to dark gray. The surrounding geopolymer binder changes its chemical composition and the observed fissures separate areas with different chemical compositions (Figures 2, 3). The EDS analyses presented in Table 6 show the chemical composition of the whole areas (surfaces A and C) along with selected points of strictly bordered areas on the pellet edges.

The geopolymer generally forms a homogenous amorphous matrix when the clayed geopolymer components are, in an ideal situation, dissolved in an aqueous alkali solution. In reality, the XRD study of the Ca–K geopolymer presented in Table 5 shows well-defined crystal phases spread in a Ca–K geopolymer matrix.

The polycrystalline character was specified and corroborated by the chemical composition of points E, F and G (Figure 3): The whitish areas are filled with the crystals, which are firmly encapsulated in the geopolymer. According to the X-ray diffraction (Table 5) the crystal phases contain sulfur oxides forming, with high probability, the arcanite.

The excess of potassium and the greater distance from the pellet edge lead to a decreased calcium content or its absence (points F and G). The migration of

| Table 5. Mineralogical compositions of the fresh ash pellets and Ca–K geopolymer matrix. |
|-----------------|-----------------|-----------------|-----------------|
| Material        | Major phases    | Minor phases    | Traces          |
| Ash pellets     | calcite (CaCO$_3$), quartz (SiO$_2$), bassanite (CaSO$_4$·0.5H$_2$O) | hematite (Fe$_2$O$_3$) | gypsum (CaSO$_4$·2H$_2$O), hydrophilite (CaCl$_2$), phengite ((K$_{0.9}$Na$_{0.05}$)(Al$_{1.51}$Mg$_{0.32}$Fe$_{0.18}$Ti$_{0.03}$)(Si$_{3.4}$Al$_{0.6}$O$_{10}$)(OH)$_2$) |
| Ca–K geopolymer matrix | quartz (SiO$_2$), calcite (CaCO$_3$), gehlenite (Ca$_2$Al(Si$_2$O$_7$)), arcanite (K$_2$SO$_4$) | | andradite (Ca$_3$Fe$_2$(SiO$_4$)$_3$) |

Figure 2. A general view of the pellets in the Ca–K geopolymer surroundings (a polished specimen).

Figure 3. A detailed view of the geopolymer with color distinction (points E, F and G).
potassium to a location farther from the pellet edge is associated with its increased mobility and potassium-sulfate crystallization. The lower calcium content is associated with its lower mobility as well as its fixation in previously formed crystal phases such as anhydrite (CaSO₄) and calcium carbonate (CaCO₃). The active forms of calcium present in the ashes such as calcium oxide and calcium hydroxide form primary ettringite (6 Ca (OH)₂·Al₂(SO₄)₃·26 H₂O) in aqueous conditions. The area C could be perceived as a specific concentration of crystals (the whiter rods in the darker surroundings in Figure 4).

A specific case is noticeable in the small area (point I) placed between the small part of the chosen pellet and its remaining base (Figure 5). The piece of the pellet was most likely separated because of its higher porosity or lower strength on the pellet edge. Point I slightly differs from the chemical composition of the pellet in area A and contains i.e. titanium. Titanium oxide is present in coal ashes; together with iron oxides, these are components of clayed substance. A different situation could be seen in the case of point J, where potassium and calcium are almost equal, but the content of sulfur is 10 times higher than in geopolymer surroundings (point H). Even the brightness is similar to that of point I; the edge point J represents a situation when the formation of sulfates is probable, detected also by the XRD and encapsulated in the geopolymer matrix. Figure 5 shows the position of point H, which is the location between the two whitish areas.

Infrared analyses

Infrared spectroscopy was used to confirm the geopolymer bonds of the final solid bodies and to define the character of the original clayed material Mefisto, the slag and the final geopolymer matrix. The resulting curves in the determinative spectral regions are presented in Figure 6. The bands at 950 - 1250 cm⁻¹ and 420 - 500 cm⁻¹ are the most intense and they are assigned to an asymmetric stretching vibration of Si–O–Si and Si–O–Al and a bending vibration of the Si–O groups. The band at 1079 cm⁻¹, as well as the 797 cm⁻¹ band of Mefisto, are associated with the asymmetric stretching of the Si–O–T group (T = Si or Al) in metakaolinite, and the band at 1079 cm⁻¹, as well as the 797 cm⁻¹ band of Mefisto, are associated with the asymmetric stretching of the Si–O–T group (T = Si or Al) in metakaolinite.

Table 6. Chemical compositions detected by EDS (wt. %).

<table>
<thead>
<tr>
<th>Element</th>
<th>C</th>
<th>O</th>
<th>Na</th>
<th>Mg</th>
<th>Al</th>
<th>Si</th>
<th>S</th>
<th>K</th>
<th>Ca</th>
<th>Ti</th>
<th>Fe</th>
</tr>
</thead>
<tbody>
<tr>
<td>Surface A</td>
<td>7.30</td>
<td>33.45</td>
<td>0.2</td>
<td>1.07</td>
<td>22.01</td>
<td>16.91</td>
<td>2.22</td>
<td>3.04</td>
<td>7.81</td>
<td>1.24</td>
<td>4.75</td>
</tr>
<tr>
<td>Point B</td>
<td>3.88</td>
<td>32.82</td>
<td>0.47</td>
<td>0.42</td>
<td>17.17</td>
<td>28.32</td>
<td>1.02</td>
<td>11.98</td>
<td>2.11</td>
<td>1.09</td>
<td>0.73</td>
</tr>
<tr>
<td>Surface C</td>
<td>5.04</td>
<td>31.35</td>
<td>0.28</td>
<td>0.54</td>
<td>10.69</td>
<td>17.96</td>
<td>9.38</td>
<td>16.95</td>
<td>7.82</td>
<td>–</td>
<td>–</td>
</tr>
<tr>
<td>Point D</td>
<td>5.67</td>
<td>24.70</td>
<td>0.18</td>
<td>–</td>
<td>4.48</td>
<td>7.34</td>
<td>15.84</td>
<td>39.13</td>
<td>0.67</td>
<td>–</td>
<td>–</td>
</tr>
<tr>
<td>Point E</td>
<td>5.08</td>
<td>30.96</td>
<td>0.41</td>
<td>0.49</td>
<td>11.95</td>
<td>20.49</td>
<td>6.64</td>
<td>22.77</td>
<td>1.20</td>
<td>–</td>
<td>–</td>
</tr>
<tr>
<td>Point F</td>
<td>5.93</td>
<td>27.26</td>
<td>0.27</td>
<td>–</td>
<td>6.42</td>
<td>10.79</td>
<td>13.65</td>
<td>35.67</td>
<td>–</td>
<td>–</td>
<td>–</td>
</tr>
<tr>
<td>Point G</td>
<td>4.95</td>
<td>25.62</td>
<td>0.14</td>
<td>–</td>
<td>3.76</td>
<td>2.25</td>
<td>18.56</td>
<td>44.73</td>
<td>–</td>
<td>–</td>
<td>–</td>
</tr>
<tr>
<td>Point H</td>
<td>13.27</td>
<td>29.8</td>
<td>0.36</td>
<td>0.70</td>
<td>14.42</td>
<td>26.75</td>
<td>0.34</td>
<td>11.14</td>
<td>3.21</td>
<td>–</td>
<td>–</td>
</tr>
<tr>
<td>Point I</td>
<td>21.49</td>
<td>29.97</td>
<td>0.16</td>
<td>2.04</td>
<td>12.69</td>
<td>17.83</td>
<td>1.70</td>
<td>1.31</td>
<td>8.74</td>
<td>0.91</td>
<td>3.15</td>
</tr>
<tr>
<td>Point J</td>
<td>8.26</td>
<td>31.95</td>
<td>0.27</td>
<td>0.60</td>
<td>8.59</td>
<td>12.03</td>
<td>11.04</td>
<td>12.91</td>
<td>12.44</td>
<td>–</td>
<td>1.89</td>
</tr>
</tbody>
</table>
where Si or Al is tetra-coordinated. However, the band at 1079 cm\(^{-1}\) can also be related to quartz. The band at 469 cm\(^{-1}\) is related to the Si–O bending vibration. The band at 912 cm\(^{-1}\) belongs to six-coordinated Al–O, but, in our case, it is hidden in the wide band at 797 cm\(^{-1}\), which can also contain another band related to quartz (778 cm\(^{-1}\)). It can be stated that Mefisto contains a large amount of Al(IV) and a lower amount of Al(VI). As arises from Figure 6, the position of the principal Si–O–T asymmetric stretching band located originally at 1079 cm\(^{-1}\) in the Mefisto sample has shifted to lower wavenumbers (1014 cm\(^{-1}\)) through a reaction with blast-furnace slag. This shift indicates an increase in the substitution of the tetrahedral Al in the silicate network due to alkaline activation and the consequently developing geopolymer network [22]. This effect can further be confirmed by the reduction in the intensity of the spectral shoulder at \(~560\) cm\(^{-1}\), attributable to silicates and/or aluminosilicates with long-range structural order in the geopolymer matrix [23]. The slag and the sample of the Ca–K geopolymer matrix contain also carbonates (875 cm\(^{-1}\)).

The deconvoluted spectrum of the Ca–K geopolymer matrix presented in Figure 7 reveals the bands belonging to metakaolinite at 1077 cm\(^{-1}\) and the weak band at 1215 cm\(^{-1}\) as the spectral shoulders of the principal band at 1070 cm\(^{-1}\). The other bands belong to Si–O and Si–O–Al at 1135 and 978 cm\(^{-1}\), respectively. As a consequence of polycondensation with alternating Si–O and Al–O bonds, the original band at 1070 cm\(^{-1}\) has shifted to lower wavenumbers and the new band at 1020 cm\(^{-1}\) for Si–O–Al asymmetric stretching has been formed [24].

The deconvoluted spectrum of pellets shown in Figure 8 reveals the bands belonging to Si–O–Al chaining at 1040 cm\(^{-1}\) [21-22, 25]. The band at 1114 cm\(^{-1}\) can be attributed to sulfates which are present in fluid fly ash and which are related to crystal phases found using the XRD method (Table 5). In this spectrum, one can also see the bands at 986 and 1174 cm\(^{-1}\), which are probably attributable to Si–O vibrations of the other silicates present in the samples as impurities.

The deconvoluted spectrum of the layer formed between the pellets and the Ca–K geopolymer matrix is presented in Figure 9. The spectrum clearly contains bands belonging to both materials, ash pellets (the bands at \(~1165\) and 1111 cm\(^{-1}\)) and the geopolymer matrix (the bands at \(~1065\), 1023 and 979 cm\(^{-1}\)). This layer forms “interspace” connecting both materials and supports the results obtained by SEM and EDS analyses.

The applied XRD analysis has shown that both matters used (the matrix and the pellets) contain roentgen-amorphous aluminosilicates and calcium-based crystal compounds in their structures. The incorporation of the pellets made from fluid fly ash into the Ca–K geopolymer matrix and the detection of the interlayer on the edges of the pellets prove the
interaction between the calcium-containing matter (the pellets – surface A) and the Ca–K geopolymer matrix. These results are supported by IR analysis, which has demonstrated that the deconvoluted spectrum of the layer formed between the Ca–K geopolymer matrix and the pellets contains bands belonging to both materials, the geopolymer matrix and the ash pellets.

CONCLUSION

The Ca–K geopolymer matrix and solidified fluid fly ash shaped into pellets (aggregates) were used for a composite preparation. It was found that these two different waste-based materials could be combined to create together specific solid, insoluble matter.

The formation of an inseparable interlayer between the Ca–K geopolymer matrix and the ash pellets has been corroborated by SEM and IR analyses. Not only are the pellets enveloped by the matrix, but their surface also reacts with the Ca–K geopolymer binder. SEM and EDS have proven the migration of calcium and potassium ions on the interface of the geopolymer matrix and the edges of the aggregates. They have also shown that the predominantly amorphous geopolymer matrix in contact with the aggregates contains well-encapsulated crystal forms of arcanite or also ettringite. A comparison of the spectra of the geopolymer matrix and the ash pellets with the spectrum of the layer formed between these materials has revealed that this interlayer contains vibrational bands that were detected in both original participants.

The water-insoluble pellets formed from the mixture of fluid fly ash and bentonite could be used as fillers or aggregates in different kinds of cementitious matter. The fabrication of larger and smaller pellets based on solidified fluid fly ash opens new ways to reuse the constantly increasing amounts of ash from coal-burning systems. Their possible combination with different waste-based materials can improve waste-material disposal.
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REFERENCES