FLEXURAL PERFORMANCE EVALUATION OF VARIOUS CARBON FIBRE FABRIC REINFORCED GEOPOLYMER COMPOSITE

HIEP LE CHI, PETR LOUDA

Department of Material Science, Faculty of Mechanical Engineering, Technical University of Liberec, Studenštíká 2, 461 17 Liberec, Czech Republic

E-mail: lechhiptul09@gmail.com

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In this paper, the results of an experimental investigation on the flexural behaviour, the mechanical properties, and the failure mode of a carbon textile reinforced geopolymer mortar composite subjected to four-point bending test are presented. The influence of various factors such as the type of the carbon fibre fabric, reinforcement ratio, dosage of the chopped basalt fibre on the flexural performance of the reinforced geopolymer composite is experimentally studied. The results reveal that the use of carbon fibre fabrics in the production of high-strength geopolymer matrix composites makes it possible for them to achieve a relatively high mechanical strength. Moreover, the value of the flexural strength and flexural toughness is strongly influenced by the reinforcement ratio and the mechanical properties of the fibre yarn of the carbon textile. The addition of the chopped basalt fibre (BF) plays an important role in both the improved mechanical strength and the failure mode of the geopolymer textile composites. The failure mode of all the specimens shows either a pure bending failure or a peeling off the concrete at the matrix/fibre interface due to the rupture of some filaments in the outer layer and the loss of the bonding strength of the fibre yarn in the matrix leading to the slippage of the fibre yarns within the matrix.

INTRODUCTION

Currently, geopolymers which are made by reacting aluminosilicate materials with an alkali activator solution have become an interesting topic in research since they offer great potential for an alternative to Portland cement-based concrete [1, 2]. It is well-known that Portland cement is a construction building material with admirable mechanical properties which are already proven. However, the Portland cement production process requires a high energy consumption and it involves a large amount of CO$_2$ emissions. This binder is attributed to one of the major causes behind global warming [3, 4]. The appearance of a geopolymer binder not only satisfies the mechanical properties, but also pays attention to a novel green building material over Portland cement in the case that there is a reasonable choice of raw materials for geopolymers [5, 6]. The eco-compatibility of geopolymers, and other peculiarities, such as its high mechanical strength, acid resistance, seawater resistance, low shrinkage, thermal conductivity and fire resistance up to 1200 °C could be successfully achieved when the key factors, such as the raw materials, mixing ratio of the fillers, the amount and concentration and type of alkaline activator solution are properly selected.

Due to the increasing need for new materials to be produced with low energy consumption that offer sustainability, economy, feasibility, and particularly good mechanical properties, textile-reinforced concrete has attracted the consideration of many scientists over the past three decades. Composites resulting from a combination of multi-axial textile reinforcement and a fine-grained aggregate-based cementitious matrix, named textile rein-forced concrete (TRC), bring the opportunity to repair/strengthen old/new reinforced concrete structures. The use of TRC as an external strengthening system has gained popularity among other techniques. This is due to the favourable properties offered by TRC, such as its resistance to corrosion, its high strength to weight ratio, its ease and speed of application and the minimum change in geometry. Furthermore, TRC composites are also considered when manufacturing lightweight concrete structures or prefabricated sandwich panels with outstanding mechanical performance, and high-performance concrete construction with free-form designs [7–15].

A textile reinforced geopolymer (TRG) is a relatively new composite material. To date, many works have been performed to evaluate the effectiveness of the use of a textile reinforced geopolymer matrix composite for
the external strengthening layer of existing structure members [16–24]. The reason for choosing geopolymer matrix composites is that they are more stable at elevated temperatures than a polymeric resin, while compared to a Portland cement matrix, they are better at anticorrosion in a chemical environment and have a lower CO₂ footprint [25]. Despite many efforts in using the TRG composites as the strengthening layers for reinforced concrete, experimental investigations on the mechanical properties of the geopolymer matrix composite reinforced with carbon fibre fabrics is still limited in the current literature.

Due to the increasing interest in the area of composite materials made by geopolymer matrix and textile reinforcements, this paper is aimed at investigating the effectiveness of a carbon fabric geopolymer matrix composite in which the matrix is a geopolymer mortar containing various dosages of a chopped basalt fibre. A proven carbon textile with high mechanical properties combined with a high-strength geopolymer matrix promises that these composite materials can be used successfully in a wide range of applications where the TRC composite is required as mentioned above, such as the retrofitting of damaged structures, the strengthening of existing structures, the pre-casting of multi-layer proof panels or tunnel linings, etc. In the previous publication, the authors presented the flexural performance of basalt fibre meshes reinforced with a geopolymer matrix [26]. The results showed that the multi-layer and small net size type of the basalt fibre mesh should be used to reinforce the specimens in order to achieve a high reinforcement effectiveness. Besides, the geopolymer mortar matrix containing a high dosage of the BF has been addressed in the previous paper [27]. However, the use of a geopolymer mortar without the BF addition displayed a mechanical strength, which was still not as high as possible; this causes anxiety whether or not the BF addition has a positive effect on the failure mode of the TRG specimens. In the current work, the geopolymer mortars with a higher mechanical strength were used as a matrix to produce the TRG specimens. Carbon textiles are one of the most popular types of textile reinforcement for composites. When compared to basalt or glass textiles, they have been found to promote a better load-bearing capacity of the reinforced composite due to the high tensile strength and Young’s modulus. Based on three types of carbon textile with respect to various net sizes, the aim of this paper is to evaluate the flexural performance of the composite thin-plates made with the high-strength geopolymer mortar in combination with carbon fibre meshes through the four-point bending test. Specimens of rectangular form with dimension of 400 × 100 × 15 mm³ were produced. The influence of some key factors such as the textile type, reinforcement ratio, dosage of the chopped basalt fibre was also studied.

**EXPERIMENTAL**

**Materials**

The geopolymer binder, supplied by Ceske Lupkove Zavody, a.s. Czech Republic, was used as the aluminosilicate source to produce the geopolymer mortar (in weight percent: SiO₂ – 47.4; Al₂O₃ – 29.7; CaO – 14.5; MgO – 2.6; TiO₂ – 1.8; Fe₂O₃ – 0.5; K₂O – 0.3; Na₂O – 1) along with a sodium silicate activator of modulus 1.73 (in weight percent: SiO₂ – 20.72; Na₂O – 12.33; H₂O – 66.68). A geopolymer cement was synthesised from calcined kaolin and shale clay residues in a rotary kiln (for 10 h at 750 °C) with a Si/Al molar ratio of 2.0.

Two different types of silica sand were used as the fine aggregate for the geopolymer mortar matrix (grain size: maximum of 0.063 mm and 0.6-1.25 mm). An anticorrosive powder additive (microsilica) based on an amorphous SiO₂ for concrete and the mortar was purchased from Kema Mikrosilika–Sanační centrum s.r.o., Sviadnov Czech Republic. The chemical composition of the microsilica was as follows (wt. %): SiO₂ – 90, CaO – 0.8, MgO – max. 1.5, Al₂O₃ – max. 1, Na₂O – 0.5. This additive was added into the geopolymer mortar to enhance both the workability of the fresh mortar and the mechanical strength of the hardened mortar. The chopped basalt fibre (BF) was provided by Kamenny Vek, and the tows were 6.4 mm long with individual fibre diameters of 13 µm, a density of 2.67 g·cm⁻³, the tensile strength was in the range of 2700 - 3200 MPa, and the tensile modulus was 85 - 95 GPa, shown in Figure 1.

Three types of the carbon fibre mesh corresponding to the net size were provided by Frisiverto s.r.o., Czech Republic, shown in Figure 1. The carbon fibre
mesh with a small net size (C-10×15) was made up of 48 K individual filaments for the yarns in the longitudinal direction and 12 K individual filaments for the yarns in the transverse direction, whereas the remaining two types (middle net size C-21×21, big net size C-34×34) were made up of 48 K individual filaments for the yarns in both directions. The yarns of the carbon textile were arranged in two orthogonal directions (0°/90°) to form a textile mesh and they were coated using a sty-ren-butadiene binder. Further detailed properties are shown in Table 1. In the four-point bending test of the geopolymer composite specimens, the carbon textiles were placed in moulds so that the force acting on the specimen was in the longitudinal direction of the fibre yarn.

Geopolymer mortar preparation and its mechanical properties

The raw materials and mixing ratio to produce the geopolymer mortar are shown in Table 2. The geopolymer mortar was prepared by the following steps. Firstly, the geopolymer cement and the activator with the given ratio were mechanically stirred for about 4 min to gain a homogenous geopolymer paste. Secondly, fine particle reinforcements, including the silica fume and micro silica sand, were added to the slurry and the mixture was stirred for about 3 min. Finally, the rough silica sand along with the chopped basalt fibre were added to the prepared mixture followed by mixing for a few minutes depending on the various BF dosage to ensure a homogenous mortar. It should be noted that due to the high content of the BF added into the mortar, in order to achieve the homogeneous mixture, the length of the BF must be changed, and it was chopped into smaller dimensions during the mixing. Thus, in this work, the original length of the BF does not affect the mechanical strength of the geopolymer mortar, but the fibre dosage does. Moreover, the mixing ratio of the particle reinforcements used in Table 2 was optimised in our lab. The freshly prepared mortar was poured into moulds with a dimension of 30 × 30 × 150 mm for the flexural and compressive tests. Three samples for each recipe were used for the flexural test and then the compressive strength was measured on both the residual pieces obtained from the flexural strength, which was conducted according to the EN 196-1 standard [28].

Table 1. The material characteristics of the carbon fibre meshes provided by the manufacturer.

<table>
<thead>
<tr>
<th>Form</th>
<th>Carbon fibre mesh</th>
<th>Carbon fibre mesh</th>
<th>Carbon fibre mesh</th>
</tr>
</thead>
<tbody>
<tr>
<td>Fibre type</td>
<td>Carbon HTC 10/15–40</td>
<td>Carbon HTC 21/21–40</td>
<td>Carbon HTC 34/34–40</td>
</tr>
<tr>
<td>Fibre density</td>
<td>1.77 g∙cm⁻³</td>
<td>1.77 cm⁻³</td>
<td>1.77 cm⁻³</td>
</tr>
<tr>
<td>Number of threads (m)</td>
<td>78 (lengthways); 55 (crossways)</td>
<td>39 (lengthways); 39 (lengthways); 26 (lengthways); 26 (lengthways);</td>
<td>26 (lengthways); 26 (lengthways);</td>
</tr>
<tr>
<td>Weight</td>
<td>350 g∙m⁻²</td>
<td>330 g∙m⁻²</td>
<td>212 g∙m⁻²</td>
</tr>
<tr>
<td>Ix</td>
<td>3200 g∙km⁻¹</td>
<td>3200 g∙km⁻¹</td>
<td>3200 g∙km⁻¹</td>
</tr>
<tr>
<td>Stitch spacing</td>
<td>10 × 15 mm² (centre to centre distance)</td>
<td>21 × 21 mm² (centre to centre distance)</td>
<td>34 × 34 mm² (centre to centre distance)</td>
</tr>
<tr>
<td>Tensile strength</td>
<td>2551 N∙mm⁻² (lengthways); 2847 N∙mm⁻² (crossways)</td>
<td>2531 N∙mm⁻² (lengthways); 2841 N∙mm⁻² (crossways)</td>
<td>2544 N∙mm⁻² (lengthways); 2720 N∙mm⁻² (crossways)</td>
</tr>
<tr>
<td>Young’s modulus</td>
<td>228 GPa (lengthways)</td>
<td>229 GPa (lengthways)</td>
<td>242 GPa (lengthways)</td>
</tr>
<tr>
<td>Elongation</td>
<td>1.17 % (lengthways)</td>
<td>1.71 % (lengthways)</td>
<td>1.47 % (lengthways)</td>
</tr>
</tbody>
</table>

Table 2. The mixture of the geopolymer mortar matrix.

<table>
<thead>
<tr>
<th>BF content (wt. % of geopolymer paste)</th>
<th>By weight ratio (-)</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td>Geopolymer cement</td>
</tr>
<tr>
<td>0, 2.5, 5, 7.5</td>
<td>1</td>
</tr>
</tbody>
</table>

Table 3. The mechanical properties of the geopolymer mortar matrix.

<table>
<thead>
<tr>
<th>Mortar type</th>
<th>Flexural strength [MPa]</th>
<th>Compressive strength [MPa]</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td>7 days</td>
<td>28 days</td>
</tr>
<tr>
<td>Without BF</td>
<td>9.46 (0.35)</td>
<td>11.78 (0.48)</td>
</tr>
<tr>
<td>2.5 % BF</td>
<td>10.87 (0.49)</td>
<td>13.34 (0.55)</td>
</tr>
<tr>
<td>5 % BF</td>
<td>11.80 (0.98)</td>
<td>14.72 (1.17)</td>
</tr>
<tr>
<td>7.5 % BF</td>
<td>12.77 (0.59)</td>
<td>14.92 (0.58)</td>
</tr>
</tbody>
</table>
Table 3 represented the flexural strength and compressive strength of the geopolymer mortar at the age of 7 and 28 days. The results show that a geopolymer mortar with a reasonable choice of mixing ratio displayed a relatively high mechanical strength and its mechanical strength increases with an increase in the BF dosage. It should also be clearly stated that a BF dosage over 7.5% can be still worked in the geopolymer mortar, however, it takes more time to mix and cast and does not indicate that it significantly improves the mechanical strength.

Manufacturing the carbon fibre fabric reinforced geopolymer composite

For the four-point bending test, the samples are moulded in a rectangular form with dimensions of 400 × 100 × 15 mm³. Carbon fibre meshes were used as the reinforcement with the geopolymer mortar matrix, and the specimens were manufactured by the hand lay-up method. The layout of carbon textile in the mould is clearly shown in Figure 2. They were positioned in the mould with a desired distance by using thin metal plates at the ends of the mould. It should be said that three reinforcing layers were the possible number in order to ensure the easy penetration of the geopolymer mortar between the textile layers when the total thickness of the specimens was 15 mm. For instance, the distance between two adjacent layers should be bigger than the size of the sand grains, so this distance was 2 mm in the case of the three reinforcing layers because the distance from the centre to the centre between the two adjacent layers was 4 mm while the self-thickness of the fibre yarn was about 2 mm. The moulds were manually vibrated for a while in order to ensure the good penetration of the fresh mortar between the textile layers after being filled with the mortar and the textile. Figure 3 shows the manufacturing process of the TRG specimens. It should be clearly stated that the number of fibre yarns per one textile layer in the bearing direction and crossways direction is: 8 fibre yarns and 24 fibre yarns for the C-10×15 mesh, 4 fibre yarns and 15 fibre yarns for the C-21×21 mesh and 3 fibre yarns and 10 fibre yarns for the C-34×34 mesh, respectively. After casting, all the specimens were wrapped using a polypropylene film, and cured at room
temperature, ~22 °C, with 45 % relative humidity for 24 h. Afterward, the specimens were demoulded, and wrapped again using a polypropylene film, and kept at room temperature until testing.

In this work, the four-point flexural performance of the textile geopolymer composites was carried out based on a different number of textile layers, textile type and various dosages of the BF. In order to evaluate the effect of the number of textile layers and the textile type, a geopolymer mortar containing 5 % BF was used. One layer of C-10×15 textile was considered to investigate the influence of the BF dosage (0 %, 2.5 %, 5 %, 7.5 %) on the flexural performance of the carbon textile geopolymer composite. Three specimens for each mixture were prepared. A total of 36 specimens were tested at approximately 40 days after the casting.

The reinforcement ratio could be defined as the ratio of the area of the textile reinforcement to the area of the specimens. It can be expressed by the following Equation 1:

$$\rho_f = \frac{A_f}{A_c}$$  \hspace{1cm} (1)

where $\rho_f$ is the reinforcement ratio, $A_f$ is the cross-sectional area of the fibre yarn of the carbon textile in mm$^2$, $A_c$ is the cross-sectional area of the TRG specimens in mm$^2$.

The cross-sectional area of the fibre yarn can be calculated by the following Equation 2:

$$A_f = \frac{\text{Tex}}{D_f}$$  \hspace{1cm} (2)

where $D_f$ is the fibre density in g·cm$^{-3}$, Tex is the linear density of the fibre yarn in g·km$^{-1}$.

Thus, the TRG specimens that are reinforced with 1-3 layers of the carbon textiles will have a reinforcement ratio of 0.96 %, 1.93 %, 2.89 % for the C-10×15 textile; 0.45 %, 0.96 %, 1.45 % for the C-21×21 textile; 0.36 %, 0.72 %, 1.08 % for the C-34×34 textile, respectively.

Test method

A four-point bending test was used to evaluate the composite performance of the TRG specimens. The detailed description of the specimen arrangement and testing process was shown in the previous publication [26]. The testing machine with a load cell capacity of 100 kN (FP Lab Test II, from LABORTECH s.r.o Opava, Czech Republic), located at the Technical University of Liberec Laboratory, with an applied load under displacement control at a loading rate of 4 mm·min$^{-1}$, was used. The four-point flexural strength in the special case where the outer support span is equal to three times the inner support span can be calculated as per Equation 3:

$$\sigma = \frac{Fl}{(bh^2)}$$  \hspace{1cm} (3)

where $\sigma$ is the four-point flexural strength in MPa; $F$ is the load at a given point on the load-displacement curve in N; $b$ is the width of the tested sample in mm; $h$ is the thickness of the sample in mm; $l$ is the support span in mm.

RESULTS AND DISCUSSION

Effect of the reinforcement ratio and textile type

The influence of the textile type and the reinforcement ratio on the flexural behaviour of the composites is shown through means of the flexural load-displacement curves in Figures 4a-c, whereas the average values of the flexural strength, the flexural toughness, the ultimate displacement and a number of cracks are described in Figures 5a-d. In general, the geopolymer composite reinforced with the carbon fibre meshes produced a displacement-hardening behaviour and the load-displacement response of these specimens varies differently which depends on the mechanical properties of the fibre yarn and the reinforcement ratio. This type of behaviour is typically characterised by three different stages, which have been clearly presented in previous research [26, 29]. The first stage is identified by a linear increase in the load until the first crack in the matrix happened. This stage is called an elastic-linear pre-cracking stage resulting from a combination of the geopolymer matrix and textile reinforcement. The second stage is a multiple-cracking phase as the proven oscillation of the curve when the applied load continues to increase, leading to the obvious pseudo-ductility behaviour of the composite. The rate of oscillation and slope of this phase depend strongly on the mechanical properties of the fibre yarn, reinforcement ratio, and mortar matrix. The number of cracks that each specimen will have is originated in this stage. The third stage is the post-cracking or crack-widening stage. In this stage,
the contribution of the geopolymer matrix is almost unaffected, the textile reinforcement governs the flexural behaviour of the composites. This stage ends once the load-bearing capacity of the composite cannot be carried out anymore through the ultimate load achieved.

The effect of the different layers of the C-10×15 textile on the flexural performance of the composites over the flexural load-displacement curves is described in Figure 4a. For the one-layer reinforced specimens, the curve response with the three different stages was clear. After reaching the maximum load-bearing capacity, the specimens were broken down smoothly characterised by the gradual decrease of the curve in the load. When using two to three reinforcing layers, the stiffness of the TRG specimens improved significantly as seen through the absence of stage 2 and the higher slope of stage 3. Using a high reinforcement ratio is one of the factors causing the absence of stage 2 which was mentioned in the previous literature [30]. The ending of stage 3 is marked by a sudden drop in the load almost to zero along with a loud noise. This was found out to be that the C-10×15 composites with the higher reinforcement ratio lead to more catastrophic brittle failure. The reason of this is that the amount of energy that the specimens absorb will be released just as much as when the specimens are destroyed; thus, the greater the amount of the accumulative energy, the more catastrophic failure of the specimens is. In this case, it can be concluded that the stiffness of the composites was governed by the stiffness of the geopolymer matrix, the material characteristic of the fibre yarn, and the reinforcement ratio. It was found that the absence of stage 2 results in the maximum flexural strength and stiffness. Similar to the C-10×15 composites, the positive effect of the different layers of the two remaining types of carbon textiles on the flexural behaviour of the TRG specimens can generally be seen in Figures 4b-c. With the increasing reinforcement ratio, the TRG specimens obviously have a higher load-bearing capacity. In other words, it should be noted that by comparing the development of stage 1 between tested specimens, the slope of first stage of the curve is nearly the same for all tested TRG specimens, but the first-crack load value does not follow a logical rule. This phenomenon is attributed to the presence of the textile reinforcement in the matrix. The stiffness and the value of first-crack load in this stage are predominantly governed by the stiffness of the matrix. However pre-existing micro-cracks, especially at the matrix/fibre interface, which can be formed during manufacturing, curing, and setting-up of the specimens, also contribute to the reduction in the first-crack load. The higher the reinforcement ratio, the greater the probability of the presence of more micro-cracks in the matrix is. On the other hand, by comparing the flexural behaviour between the C-21×21 composites and the C-34×34 composites, it was found that the mechanical properties of the fibre yarn have a strong effect on the response of the flexural load-displacement curves which leads to the interesting results of the mechanical properties of the TRG composites (Figure 5). As mentioned in Table 1, although the tensile strength of the fibre yarns has similar values (only considering the fibre yarns in the load-bearing direction), the fibre yarns of the C-21×21 textile show a lower Young’s modulus and higher elongation than those of the C-34×34 textile; thus, as a result, the C-21×21 composites will tend to exhibit a more pseudo-ductile behaviour in bending, compared to those reinforced with the C-34×34 textile, as proven by the lower slope and less oscillation in stage 2. In addition, in stage 3, just before and just after reaching the maximum load-bearing capacity, the C-21×21 composites almost maintain the unchanged load-bearing capacity with an increasing displacement (Figure 4b).
The effect of the reinforcement ratio and the textile type on the flexural strength, the flexural toughness, the displacement and the crack number of the TRG specimens is presented in Figures 5a-d. It should be noted that due to using the geopolymer matrix with the same recipe for all the tested specimens, the different comparison of the first-crack strength between the tested specimens is not mentioned, and the difference in the first-crack strength observed between the tested specimens was discussed above. In general, the mechanical strength improved significantly with the increasing reinforcement ratio. The superiority of the reinforcement ratio in the C-10×15 composites makes them achieve a remarkably higher flexural strength, compared to those reinforced with the C-21×21 textile and the C-34×34 textile (Figure 5a). Even the composites reinforced with three layers of the C-21×21 and C-34×34 textiles have almost the similar flexural strength achieved by the C-10×15 composites reinforced with one-layer only. When comparing the flexural strength between the C-21×21 composites and the C-34×34 composites, the mechanical properties (Young’s modulus and elongation) of the fibre yarn have a significant influence on the composite strength. Although the C-21×21 composites apparently have a higher reinforcement ratio than the C-34×34 composite (considering the same reinforcing layer), they could not produce a higher value in the flexural strength. For the C-10×15 textile, the specimens reinforced with two and three layers increased in the flexural strength by 58.42 % MPa, 113.83 % MPa, respectively, compared to the one-layer reinforced specimens. For the C-21×21 textile and the C-34×34 textile, the specimens reinforced with two and three layers increased in the ultimate flexural strength by 38.79 %, 45.12 % and 117.73 %, 102.62 %, respectively. Figure 5b shows the results of the flexural toughness of the TRG specimens. The flexural toughness, the so-called absorption energy capacity, is defined as the resistance of a material to failure or crack propagation. Toughness implies the amount of energy the specimen is able to absorb under loading up to a

![Figure 5a](image1.png)

![Figure 5b](image2.png)

![Figure 5c](image3.png)

![Figure 5d](image4.png)

Figure 5. The experimental results of the TRG composites with respect to the textile type and the different number of textile layers: a) the ultimate flexural strength; b) the flexural toughness; c) the ultimate displacement; d) the number of cracks.
specified displacement, which is clearly described in the ASTM/C 1609 Standard. In this work, it is measured as the area under the respective load-displacement curves up to the peak load (ultimate load). It was observed that when the specimens reinforced with three layers of the textile are considered, the C-21×21 composites show the highest value of the flexural toughness (Figure 5b). It can be explained by fact that the high elongation of the C-21×21 textile positively affected the TRG composite ductility (Figure 5c), which compensates for the low flexural strength; as a result, the flexural toughness was improved. This reveals that using both a reasonable reinforcement ratio and textile type make the composites possible to significantly enhance the flexural toughness. Figure 5d shows that the number of the cracks increased with the increasing number of the textile layers. Based on this result, we also see that the stiffness of the composites is related to the number of cracks. The composites with the higher stiffness will delay the propagation of the cracks longer, which results in the reduction in the crack width and an increase in the number of the cracks at the tension edge.

The flexural failure modes of the TRG composites are displayed in Figure 6. It was observed that the TRG specimens could exhibit two different types of bending failure: i) a loss of the bonding strength at the matrix/fibre interface along with a partial rupture of the filaments at the outer layer leads to the slippage of the fibre yarns within the mortar matrix resulting in a pure flexural failure, ii) the initiation of the failure process is the same like the type 1 failure mode, then as soon as the TRG specimens reach their maximum load-bearing capacity, a sudden debonding process and a catastrophic collapse occurs almost simultaneously with loud noises. As seen in Figure 6, almost all of the TRG specimens have the same type 1 failure mode, while those reinforced with 2-3 layers of the C-10×15 textile have the type 2 failure mode.

Effect of the chopped basalt fibres

The effect of the dosage of the chopped basalt fibres on the flexural behaviour of the TRG specimens reinforced with one layer of the C-10×15 textile is described through means of the flexural load-displacement curves in Figure 7, whereas Figure 8a-b show the mean values of the flexural strength, the flexural toughness, and the number of cracks of the corresponding specimens. It can be seen that the mortar matrix with the BF addition significantly improved the mechanical properties of the TRG composites. The positive effects of increasing BF dosage on the load-displacement responses of the composites can be clearly seen through a higher first-crack load, higher maximum bearing capacity, less fluctuation in stage 2 and higher stiffness, compared to those without the BF addition (Figure 7). The improvement in the mechanical
properties of the textile-reinforced composite resulting from the BF addition could be explained as follows: i) the randomly distributed fibres in the matrix will help to reduce the overall shrinkage strains, thus decreasing the internal flaws in the matrix. On the other hand, if they do occur, the fibres could reduce their development which induces the formation of micro-cracks [31]; ii) due to the bridge effect of the chopped fibre at the micro-cracks, the mortar matrix has higher mechanical properties since the shift from micro-cracks to macro-cracks demands higher stress. Moreover, the connection between the fibre yarns of the textile and the geopolymer mortar is also promoted due to this bridging effect. In this case, it could be said that the chopped fibre contributes to enhancing the efficiency of the textile in the reinforcement as well. The TRG composites with the BF addition confirm the high-quality advancement in the aspect of the bending strength and, the higher the fibre dosage, the higher the bending strength when compared to the composites without the BF addition. The average ultimate flexural strength of the TRG specimens without the BF addition is 28.64 MPa. The average ultimate flexural strength of the TRG specimens with the BF addition of 2.5 %, 5 %, and 7.5 increases by approximately 19.62 %, 40.92 %, and 52.89 %, respectively, compared to those without the BF addition. The specimens also increase in the flexural toughness with the increasing BF content. The average flexural toughness of the specimens without the BF addition is 21.03 kN∙mm. The average flexural toughness of the specimens with the BF addition of 2.5 %, 5 %, and 7.5 %, increases by 7.04 %, 32.00 %, and 46.83 %, respectively, compared to those without the BF addition. In addition to improving the mechanical properties, the BF addition also clearly impacts the failure mode of the TRG specimens, as seen in Figure 9. There are two different types of failure modes which demonstrate the general failure modes of all the TRG specimens. For the TRG specimens without the BF addition, the poor bonding performance between the textile layer and the geopolymer matrix leads to the gradual peeling process of the fibre yarns that occurs out from the matrix, followed by the collapse of the matrix due to it reaching its maximum load-bearing capacity. It was found that although the geopolymer mortar without the BF addition used in this work had significantly higher mechanical strength than that in the previous publication [27], both types of mortar still induce the TRG specimens to have the same failure mode as the debonding phase within the matrix occurred. The textile reinforced specimens with the BF addition, on the contrary, result in a pure flexural failure without a debonding phase due to the slipping of the fibre yarn within the matrix. It can be said that the use of chopped basalt fibres in the production of the geopolymer mortar has markedly improved the rigidity of the matrix structure and has also promoted the cohesion.
of the fibre yarns of the carbon textile in the matrix. This result confirms that adding the BF into geopolymer matrix positively impacts the failure mode of the TRG specimens, and the use of the one-layer textile reinforcement is not strong enough to peel off the mortar matrix.

The relationship between the reinforcement ratio and flexural strength was demonstrated in Figure 10, where the graph was generated by 36 tested TRG specimens in total. It was observed that the flexural strength of the TRC specimens are a function of the reinforcement ratio. Although there is a marked difference in the flexural strength between the TRG specimens due to the difference in the textile type and dosage of the BF, a good fit is still found (0.90) which confirms the relatively good relationship between the flexural strength and the reinforcement ratio.

Figure 11 shows the representative SEM micrographs of the fibre yarns of the carbon textile extracted from the TRG specimens after the bending test. It can be clearly seen that some of the outer layer filaments are brittle and it seems that the mortar matrix containing the BF induces a greater number of broken fibre filaments. This result can be concluded in the fact that the addition of the BF is a key factor in improving the bonding in the TRG composites. This evidence is consistent with the improved flexural strength and failure mode discussed above.

Figure 10. The correlation between the reinforcement ratio and the flexural strength.

Figure 11. The SEM images of the carbon fibre yarns extracted from the TRG specimens after the bending test.

a) Carbon fibre yarn embedded in the geopolymer mortar without the BF addition

b) Carbon fibre yarn embedded in the geopolymer mortar with the addition of the 5 % BF
CONCLUSIONS

Based on the three types of carbon fibre fabric corresponding to the net size provided by Frisiverto s.r.o., Czech Republic, this paper experimentally investigated the flexural performance of the carbon textile reinforced geopolymer composite. The TRG specimens were produced and tested at the age of about 40 days after casting to evaluate the influence of the reinforcement ratio, carbon fabric type and various dosages of the BF on the mechanical properties of the TRG specimens. Based on the experimental results achieved, the following conclusions can be pointed out:

- With the increasing reinforcement ratio, the flexural strength of the TRG specimens were improved significantly especially with the TRG specimens reinforced with the C-10×15 textiles. In other words, the TRG specimens reinforced with the one-layer of the C-10×15 textile have an almost similar flexural strength achieved by those reinforced with the three layers of the C-21×21 textile and the C-34×34 textile.

- Using a high reinforcement ratio makes the C-10×15 composites achieve a high flexural strength, but they were catastrophically destroyed when compared to the remaining TRG specimens. The debonding and collapse phase took place simultaneously.

- Although the C-21×21 composites have a higher reinforcement ratio (considering the same reinforcing layer), they are unable to produce a higher flexural strength than those reinforced with the C-34×34 textiles.

- In contrast to the flexural strength, the C-21×21 composites reinforced with the three layers (a reinforcement ratio of 1.45 %) achieved the highest flexural toughness while the C-10×15 composite reinforced with three layers (2.89 %) achieved the lowest one. The reason for this is that the higher elongation of the C-21×21 textiles makes the TRG specimens obviously achieve greater ductility, which offsets their low flexural strength; as a result, the C-21×21 composites have a greater flexural toughness.

- The results obtained in this experiment suggest that the C-10×15 composites are suitable for high-strength and low-ductility applications while the composites reinforced with the remaining two types of carbon textile are useful for high-ductility applications.

- The chopped basalt fibres that were added into the geopolymer mortar were seen to significantly improve the mechanical properties and failure mode of the TRG specimens. Moreover, the BF addition will help to reduce the overall shrinkage strains, thus decreasing the internal flaws in the matrix; as a result of improving the mechanical strength of the TRG specimens.

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REFERENCES


