

MICROWAVE-HYDROTHERMAL SYNTHESIS AND PHOTOCATALYTIC PROPERTIES OF Cr-DOPED SrTiO₃ POWDERS

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The synthesis of Cr-doped SrTiO₃ ($x = 0, 0.02, 0.04$ and 0.1) was performed by the microwave-hydrothermal method. The fast synthesis of SrTi_{0.96}Cr_{0.04}O₃ pure phase powders was achieved at 200 °C for 30 min. It was observed that chromium entered into the solid solution within the SrTiO₃ lattice as was demonstrated from the XRD and Raman analyses. The mean crystal size of the synthesised powders was close to 60 nm as determined by XRD analysis and confirmed by the TEM images. The specific surface was measured and the values range from 27 ± 2.8 to 37 ± 3.7 m²·g⁻¹. The photocatalytic properties were evaluated for the hydrogen evolution from the methanol solution. The best performance was observed for $x = 0.04$, for which one of the highest surface areas was obtained (34 m²·g⁻¹) among the investigated compositions.

INTRODUCTION

Research on photocatalytic materials for water splitting and wastewater remediation has increased due to environmental concerns. The production of clean energy is being urged to cover the increasing energy demand and to avoid the use of fossil fuels that produce toxic gases during its burning. On the other hand, to achieve this goal, it is desirable to develop materials that can use sunlight (visible spectra) for this purpose, as it is an inexpensive and clean energy source. Among the prospect materials which can fulfil most of the requirements for water splitting are some semiconductor ceramics with a perovskite structure [1-2]. One of these materials is strontium titanate (SrTiO₃ or STO) that is known to possess a photocatalytic activity [3-5]. Unfortunately, STO has a wide bandgap (~ 3.2 eV), so that it is only active under UV-light irradiation. Then, in order to develop materials active under visible light irradiation, metal cations have been introduced into the STO lattice, lowering the absorption edge to the visible light region (red shifting). The majority of the research has been conducted using transition metal cations, noble metal cations and some non-metal anions [6-10]. The results

have shown a reduction in the bandgap and an increase in the photocatalytic response. In addition to the doping strategy for enhancing the catalytic response, crystal size and morphology have also been modified in an attempt to improve the photocatalytic performance of STO. To achieve these modifications on the STO features, several synthesis methods have been tested, including a solid-state reaction [3, 11-12], sol-gel [13-14], Pechini [6, 15], conventional hydrothermal [16-17], solvothermal [4, 18-19], electrospinning [20], microwave-hydrothermal [21-22] and flux methods [23]. Most of these techniques require high temperatures or a long heat treatment time for the powder crystallisation. The microwave-hydrothermal method has recently gained popularity due to the low temperature required, fast synthesis time and the possibility to manipulate the crystal size and shape of the prepared powders [21-22, 24-25], along with the high purity in the produced materials. As a result, the properties of the synthesised powders can be manipulated according to the desired features.

In this work, SrTi_{1-x}Cr_xO₃ ($x = 0, 0.02, 0.04$ and 0.1) powders were synthesised by the microwave-hydrothermal method and their photocatalytic properties were evaluated in the H₂ evolution.

EXPERIMENTAL

Synthesis of powders

For the synthesis of the STO and Cr-doped STO ceramics ($\text{SrTi}_{1-x}\text{Cr}_x\text{O}_3$, $x = 0, 0.02, 0.04$ and 0.1) Titanium isopropoxide ($\text{Ti}[\text{OCH}(\text{CH}_3)_2]_4$, Sigma-Aldrich 97 %), Strontium nitrate ($\text{Sr}(\text{NO}_3)_2$, Sigma-Aldrich 99 %), Chromium(III) nitrate nonahydrate ($\text{Sr}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$, Sigma-Aldrich 99 %), Potassium hydroxide (KOH, 2 M), Isopropanol (JT Baker) and deionised water were used as the starting reagents. In a typical experiment, strontium and chromium nitrates were weighed according to the desired composition, and added into the reaction vessel followed by the addition of 8 ml of isopropanol under constant stirring. Then titanium isopropoxide was added and stirred for 15 min. Afterwards, 7 ml of deionised water were mixed with the original solution producing a gel, to which 10 ml of KOH 2M were added, and the resultant suspension was stirred for 15 minutes. The final mixture was sealed in a Teflon lined autoclave and heat treated at 200 °C for 60 minutes. To further investigate the possibility of reducing the synthesis time of the pure crystalline powders, experiments were conducted at 30 minutes for the composition of $x = 0.04$. After the reaction time, the resultant precipitate was filtered, washed three times with deionised water and dried at 120 °C for 3 h.

Structural and microstructural characterisation

The crystal structure was determined by X-ray powder diffraction (XRD) with a D2-Phaser diffractometer (Bruker) using $\text{Cu K}\alpha$ ($\lambda_1 = 1.5406 \text{ \AA}$, $\lambda_2 = 1.5444 \text{ \AA}$, λ_2/λ_1 ratio = 0.5) radiation. The data were acquired from 15° to 100° in the 2θ range. Evaluation software of the XRD patterns (EVA, Bruker) was used for the crystal mean size calculation. The powder morphology was observed by scanning electron microscopy (SEM) with a JEOL JSM IT300 microscope, also, transmission electron microscopy (TEM) was used to observe the features of the crystals (JEOL JEM ARM200F). The chemical composition analyses were performed with the Energy Dispersive X-Ray Analysis (EDX) methodology with 5 minutes of exposition to ensure good statistics. Raman spectroscopy can be used to monitor changes in the lattice response to doping. Then, the Raman analysis was performed with Bruker Senterra II equipment ($\lambda = 532 \text{ nm}$, 5 mW) from 50 to 1200 cm^{-1} .

Optical absorption and band gap calculation

The absorption spectra were obtained by using the diffuse reflectance technique. The measurements were performed in a Cary 5 spectrophotometer equipped with a Praying Mantis (Harrick Scientific Products, Inc.)

accessory for the diffuse reflection spectroscopy. The spectra were recorded in the wavelength range between 200 and 800 nm. The band gaps were estimated from the absorption spectra following the procedure described elsewhere [16].

Specific surface area and photocatalytic performance

In order to measure the specific surface area, the Brunauer–Emmett–Teller (BET) method was used with Quantachrome Autosorb 3B equipment. Before the surface analysis, the samples were heated at 250 °C for 5 h for degasification.

The photocatalytic performance was evaluated by quantifying the H_2 evolution. In a typical experiment, 50 milligrams of powder were loaded into a water-methanol solution (1:1 ratio). To ensure the adsorption-desorption equilibrium, the system was stirred in the dark for 15 min and degassed by bubbling nitrogen, prior to the photochemical reaction test. The water splitting reaction was undertaken in a quartz closed reactor cell coupled with a gas chromatograph (Shimadzu G-08) equipped with a thermal conductivity detector (TCD) and a Shincarbon packed column (2 m length, 1 mm ID and 25 mm OD), using N_2 as the carrier gas. The amount of evolved H_2 was measured at time intervals of 1 h for a time period of 5 h. The experiments were performed with UV radiation (254 nm) produced by a high-pressure mercury lamp (Pen-Ray Mercury Lamp with $I_0 = 2 \text{ mW}\cdot\text{cm}^{-2}$) and without any loaded cocatalyst.

RESULTS AND DISCUSSION

In Figure 1, the XRD patterns of the undoped and Cr-doped SrTiO_3 powders are shown. For all the compositions, a pure phase is observed when the samples were synthesised at 200 °C for 60 min. The $\text{SrTi}_{0.96}\text{Cr}_{0.04}\text{O}_3$ composition was also prepared reducing the synthesis time to 30 min. The results presented in Figure 1b also suggest that the pure phase was observed for samples obtained under a reduced synthesis time. This could be achieved due to the high energy input using the microwave-hydrothermal method (MHM). The activation energy does not change when a reaction takes place, but using the MHM, the energy input rate is increased, which makes the high synthesis rate of materials possible [24]. The fast synthesis of other materials has been reported by the MHM including NaNbO_3 , $\text{K}_{0.5}\text{Na}_{0.5}\text{NbO}_3$, BiFeO_3 and KNbO_3 [25-27].

On the other hand, the preparation of SrTiO_3 has been reported using conventional hydrothermal methods with a reaction time exceeding 7 h [16]. While Cr–La-doped SrTiO_3 materials have been synthesised by the MHM method, with a synthesis time of 90 min [22]. Moreover, the solvothermal method has been used for the preparation of Cr-doped SrTiO_3 , but with a longer synthesis time and an additional calcination treatment [4].

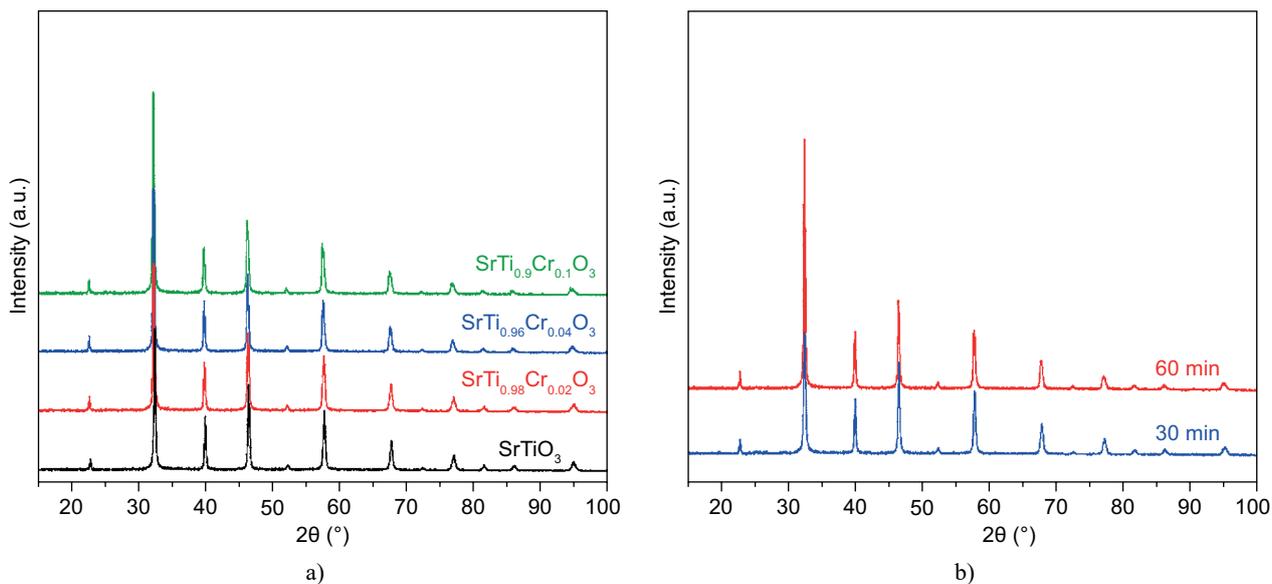


Figure 1. XRD patterns of the undoped and Cr-doped SrTiO₃ powders synthesised under different conditions: a) at 200 °C and 60 min, b) SrTi_{0.96}Cr_{0.04}O₃ at 200 °C for different times.

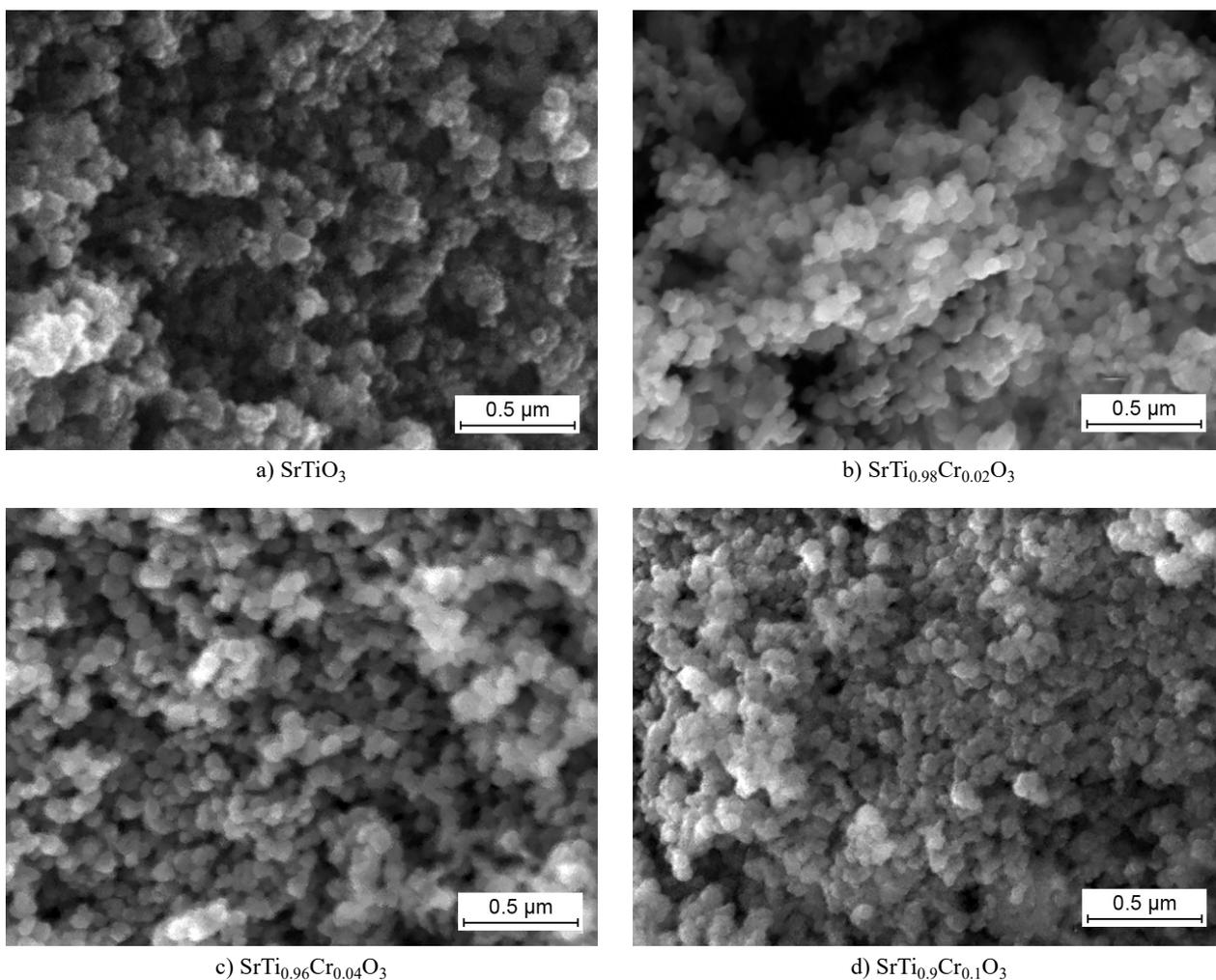


Figure 2. SEM images of the undoped and Cr-doped SrTiO₃ powders synthesised at 200 °C and 60 min: a) SrTiO₃, b) SrTi_{0.98}Cr_{0.02}O₃, c) SrTi_{0.96}Cr_{0.04}O₃ and d) SrTi_{0.9}Cr_{0.1}O₃.

From the SEM images shown in Figure 2, it is possible to observe that all the samples are composed of nanometric crystals. This feature would have a positive effect on the photocatalytic performance, because the distance for the electron-hole pair migration path is reduced. Furthermore, the specific surface area is high in nanometric powders which increase the reaction sites [28]. Regarding the average crystal size obtained in Cr-doped STO powders, the measured values using the XRD results were: 57.1 ± 1.0 , 41.0 ± 0.8 , 41.5 ± 0.8 , 37.5 ± 0.4 , for $x = 0, 0.02, 0.04$ and 0.1 , respectively. Similar results, with values close to 30 nm, were obtained for related compositions using the conventional hydrothermal method plus calcination [17] and the microwave-hydrothermal technique [22]. The TEM images of the $\text{SrTi}_{0.96}\text{Cr}_{0.04}\text{O}_3$ sample synthesised at 200 °C for 60 min can be observed in Figure 3. For the better resolution of the powder morphology, the bright and scanning transmission electron microscopy (STEM) modes were used. From these images, the cubic-like morphology of the prepared powders is clearly observed. This and other images were used to measure the mean crystal

size of $62 \text{ nm} \pm 11 \text{ nm}$, which, for this purpose, around 130 crystals were considered. This experimental value agrees with the results obtained from the XRD analysis (considering the error in the measurement) and the SEM images presented above.

In Table 1 the energy-dispersive X-ray spectrometry (EDS) analyses for the undoped and $x = 0.1$ compositions are presented. Both analyses demonstrate that the pure phase is obtained in the Cr-doped STO powders. The results are practically the nominal compositions considering the errors ascribed to the technique, and those results are reliable enough due to the new technology development for the EDS measurements [29-30].

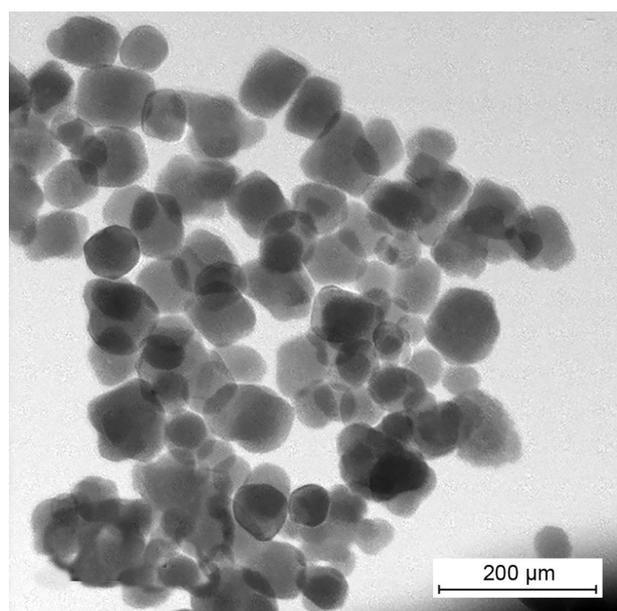
The results for the band gap measurement are shown in Figure 4 and summarised in Table 2. From these data, band gap values between 3.28 and 2.18 eV were obtained for the undoped and Cr-doped STO powders. These values are comparable with those reported in literature for similar compositions [16,19,20,22,31]. For example, Sulaeman et al. [19] reported band gap values for $x = 0$ and $x = 0.02$ of 3.18 and ~ 2.2 eV, respectively, while Akita et al. [22] reported for $x = 0$ and $x = 0.3$ (Cr-La co-doped STO) band gaps of 3.28 eV and 2.35 eV, respectively.

Table 1. Chemical analysis by EDS of the undoped and 10 mol. % Cr-doped powders.

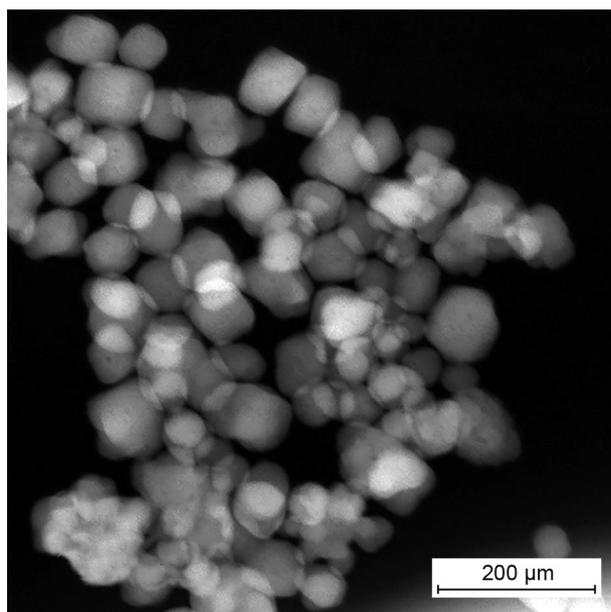
Composition	0 %	10 %
Element	Atomic %	Atomic %
O	66.65 ± 0.61	68.99 ± 0.60
Sr	17.90 ± 0.08	15.02 ± 0.06
Ti	15.44 ± 0.12	14.57 ± 0.06
Cr	0	1.42 ± 0.06
Total:	100.00	100.00

Table 2. Band gap of the Cr-doped powders.

Sample	Band gap
$\text{SrTiO}_3 - 200 \text{ °C}/60 \text{ min}$	3.28 eV
$\text{SrTi}_{0.98}\text{Cr}_{0.02}\text{O}_3 - 200 \text{ °C}/60 \text{ min}$	2.48 eV
$\text{SrTi}_{0.96}\text{Cr}_{0.04}\text{O}_3 - 200 \text{ °C}/60 \text{ min}$	2.40 eV
$\text{SrTi}_{0.9}\text{Cr}_{0.1}\text{O}_3 - 200 \text{ °C}/60 \text{ min}$	2.37 eV
$\text{SrTi}_{0.96}\text{Cr}_{0.04}\text{O}_3 - 200 \text{ °C}/30 \text{ min}$	2.40 eV
$\text{SrTi}_{0.96}\text{Cr}_{0.04}\text{O}_3 - 180 \text{ °C}/30 \text{ min}$	2.18 eV



a) Bright field



b) STEM

Figure 3. STEM images of the $\text{SrTi}_{0.96}\text{Cr}_{0.04}\text{O}_3$ sample synthesised at 200 °C for 60 min with the microwave-hydrothermal method: a) Bright field and b) STEM.

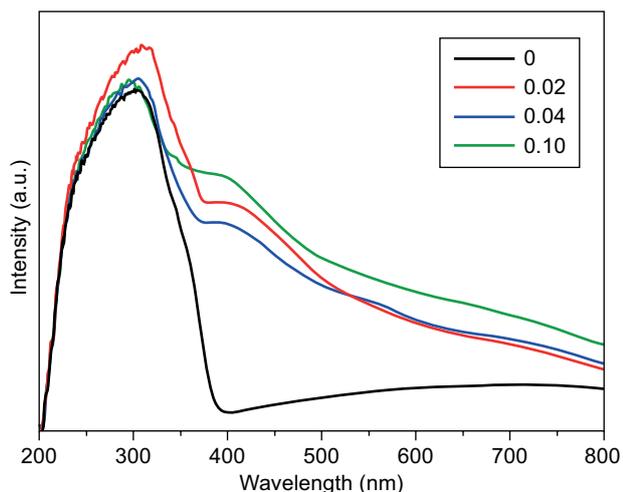


Figure 4. Diffuse reflectance spectra for the Cr-doped powders synthesised at 200 °C and 60 min.

The addition of chromium has promoted a decrease in the band gap due to the intermediate electronic levels introduced. This is clearly evident because the colour of the powders has changed from white for the STO to dark-green for the Cr-doped samples. From the UV-vis graph, it is evident that the band gap has shifted to the visible region. The absorption edge has changed from near 400 nm to 450 - 490 nm, corresponding to the undoped and Cr-doped samples. The absorption band that appears close to 460-500 nm is ascribed to the Cr³⁺-Ti⁴⁺ charge transfer. The intermediate level introduced by the Cr³⁺-doping produces a hybridisation in the conduction band that is constructed by the Ti⁴⁺ 3d and Cr³⁺ 3d orbitals [22,31].

The Raman spectroscopy results are presented in Figure 5. For the undoped sample, the spectra have several vibrations signals, ranging from 80 to 1024 cm⁻¹. These results agree with previous reports for the undoped STO [21,32]. The broad anomalies close to 300 and

700 cm⁻¹ correspond to first and second order vibrations [32]. This means that there are local compositional fluctuations that represent a phase transition to a polar phase. Furthermore, the 770 cm⁻¹ signal is ascribed to the disorder due to the addition of Cr³⁺ cation [32], which is why it is not seen for the undoped sample. Moreover, this signal becomes broader and shifts slightly to a higher wavenumber with the increasing Cr³⁺ doping (Figure 5b). Furthermore, the 175 and 275 cm⁻¹ anomalies are ascribed to the O-Sr-O bending modes, while that one at 545 cm⁻¹ is assigned to the Ti-O-Ti bending mode [21]. Then, the Raman spectroscopy is capable of sensing the doping in solid solutions.

According to the above results, the purity, crystal size and band gap requirements for adequate water splitting are fulfilled. Then a high photocatalytic activity is expected for powders synthesised by the microwave-hydrothermal method.

The specific surface area and photocatalytic performance for the hydrogen production are shown in Table 3. Generally speaking, the specific surface area increases slightly with the Cr-doping, and the highest value is reported for $x = 0.1$ which agrees with the decreasing trend in the measured average crystal size.

Regarding the photocatalytic properties, the hydrogen production also increases with the Cr-doping. As stated above, the band gap has been reduced due to the intermediate electronic levels introduced by the chromium, and, as a result, the hydrogen production has increased. Furthermore, the increase in the surface area also may have influenced the H₂ evolution. The catalytic performance depends upon parameters such as the surface area, band gap, and crystallinity [31]. Then, considering these issues, it is observed that specific surface area has played an important role in increasing the H₂ generation, because the band gap values are similar for most Cr-doped SrTiO₃ synthesised compositions and

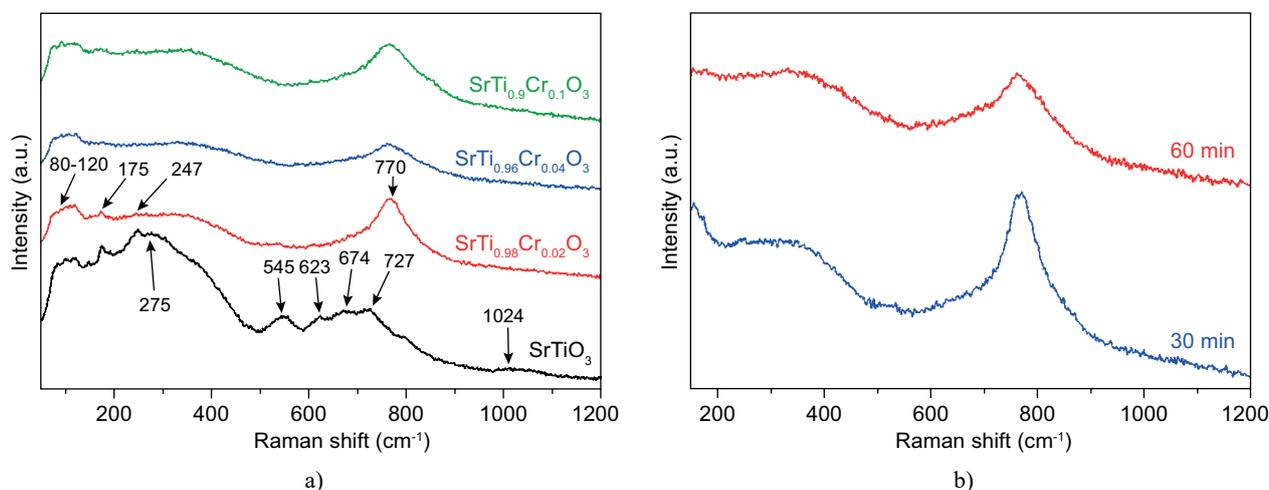


Figure 5. Raman spectra of the Cr-doped powders synthesised at: 200 °C and 60 min and b) SrTi_{0.96}Cr_{0.04}O₃ at 200 °C and different times.

Table 3. Photocatalytic activity of the Cr-doped powders.

Sample	Specific surface area ($\text{m}^2\text{-g}^{-1}$)	H_2 production ($\mu\text{mol}\cdot\text{h}^{-1}\cdot\text{g}_{\text{cat}}^{-1}$)
$\text{SrTiO}_3 - 200\text{ }^\circ\text{C}/60\text{ min}$	27 ± 2.8	438 ± 21
$\text{SrTi}_{0.98}\text{Cr}_{0.02}\text{O}_3 - 200\text{ }^\circ\text{C}/60\text{ min}$	27 ± 2.8	498 ± 24
$\text{SrTi}_{0.96}\text{Cr}_{0.04}\text{O}_3 - 200\text{ }^\circ\text{C}/60\text{ min}$	28 ± 2.9	409 ± 20
$\text{SrTi}_{0.9}\text{Cr}_{0.1}\text{O}_3 - 200\text{ }^\circ\text{C}/60\text{ min}$	37 ± 3.7	408 ± 19
$\text{SrTi}_{0.96}\text{Cr}_{0.04}\text{O}_3 - 200\text{ }^\circ\text{C}/30\text{ min}$	34 ± 3.5	504 ± 25
$\text{SrTi}_{0.96}\text{Cr}_{0.04}\text{O}_3 - 180\text{ }^\circ\text{C}/30\text{ min}$	33 ± 3.4	483 ± 24

the heat treatment was similar. Furthermore, this could explain some of the results in Table 3. For example, samples with $x = 0.02$ synthesised for 60 min has a similar photocatalytic performance to the sample with $x = 0.04$ processed during 30 min. In these two samples, as the reaction time is reduced, a smaller crystal size is observed and a higher surface area is obtained, which benefits the photocatalytic activity [6]. The sample with $x = 0.1$ presents the highest surface area among the prepared samples, but the activity is the lowest, this could be related to an increase in the dopant concentration that increases the defects (vacancies and lattice distortions), which function as electron-hole recombination centres [20].

On the other hand, the H_2 evolution reported in this work is higher compared with previous reports [19,31]. This could be explained by the low synthesis temperature, higher crystallinity and high surface area obtained in the Cr-doped STO by the microwave-hydrothermal method.

In relation to the H_2 evolution, this increases with time and this performance was stable during the 5 h that the experiments lasted. Moreover, the powders maintained their structural stability during the measurements.

CONCLUSIONS

Pure phase Cr-doped SrTiO_3 powders were synthesised by the microwave-hydrothermal assisted method under reduced time conditions. The crystal mean size for the different compositions measured from the XRD patterns ranged from 37.5 ± 0.4 to 57.1 ± 1.0 nm. The band gap was effectively reduced from 3.28 eV for the undoped SrTiO_3 to 2.36 eV for $x = 0.1$. A high surface area was obtained in the crystalline powders, with the highest values for $x = 0.1$. A good H_2 evolution was measured for all the samples, with a higher activity observed for the $\text{SrTi}_{0.96}\text{Cr}_{0.04}\text{O}_3$ composition. It was observed that a combination of energy levels was introduced by the Cr-doping (diminishing band gap), as well as the increasing the surface area (short reaction time that produce nanometric powders) plays a crucial role in order to enhance the H_2 evolution using Cr-doped SrTiO_3 materials.

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